

## Tools for

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# Low Frequency Gravitational Wave Interferometric Detectors

Glassy Metal flex joints

Are G.M. **better than fused silica**  
for mirror suspensions?

Riccardo DeSalvo et al.

6th of September 2002

LIGO-G020445-00-R

# The Glassy Metals Team

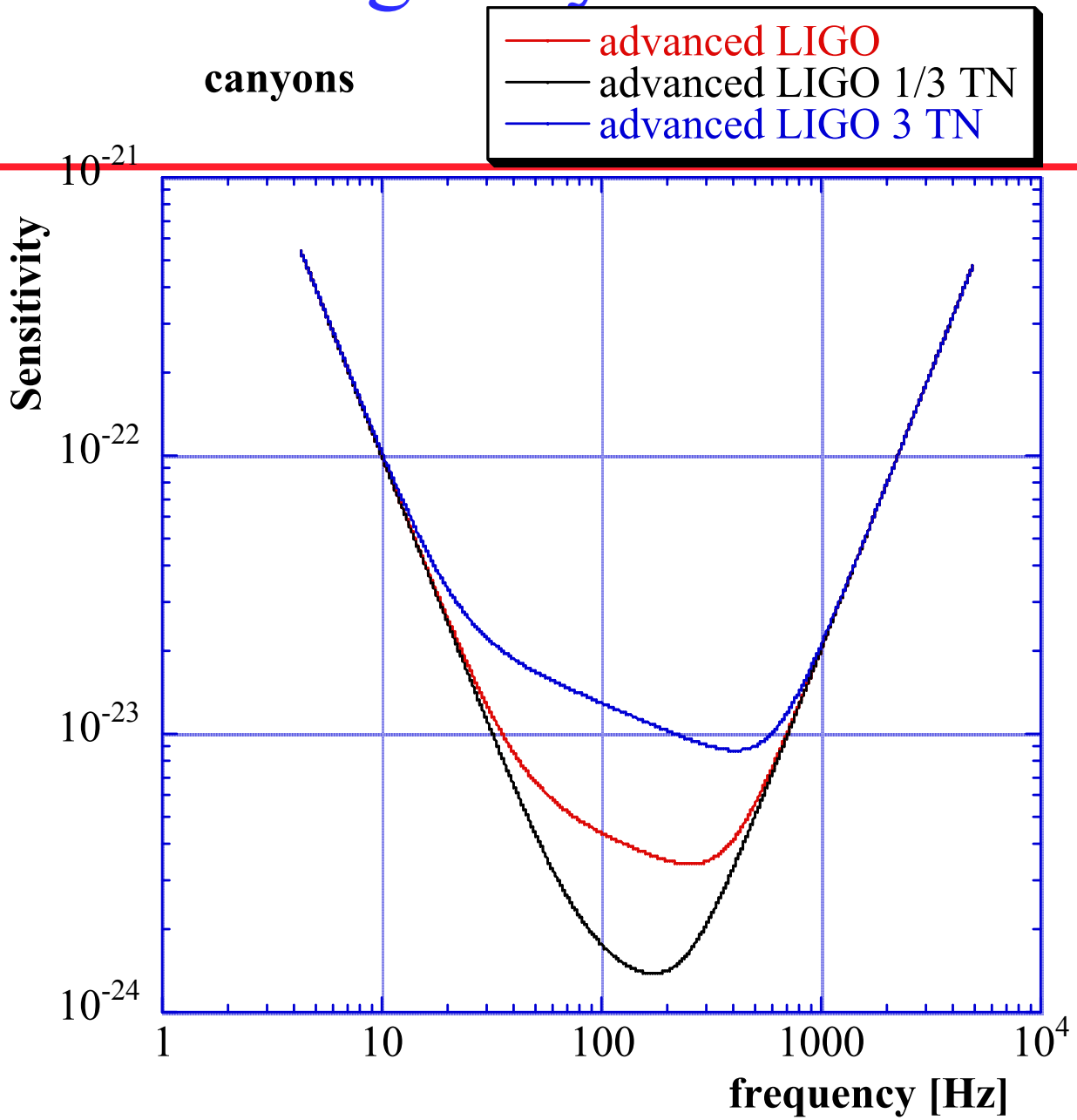
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- Maddalena Mantovani
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  - Brian Emmerson
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  - Thermal Properties / Undergraduate
  - Data Analysis (TP) / HS Student
  - Creep Analysis / Undergraduate
  - FEA of Flex Joint / Undergraduate
  - Sample Uniformity / Undergraduate
  - FEA of Flex Joint / Undergraduate
  - Phase Transition / Undergraduate
  - Stress & Strain / Graduate Student
  - Coordinator / Graduate Student
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  - Professor (Materials Science)
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# Narrowing canyons

Any improvement of thermal noise narrows the sensitivity canyon

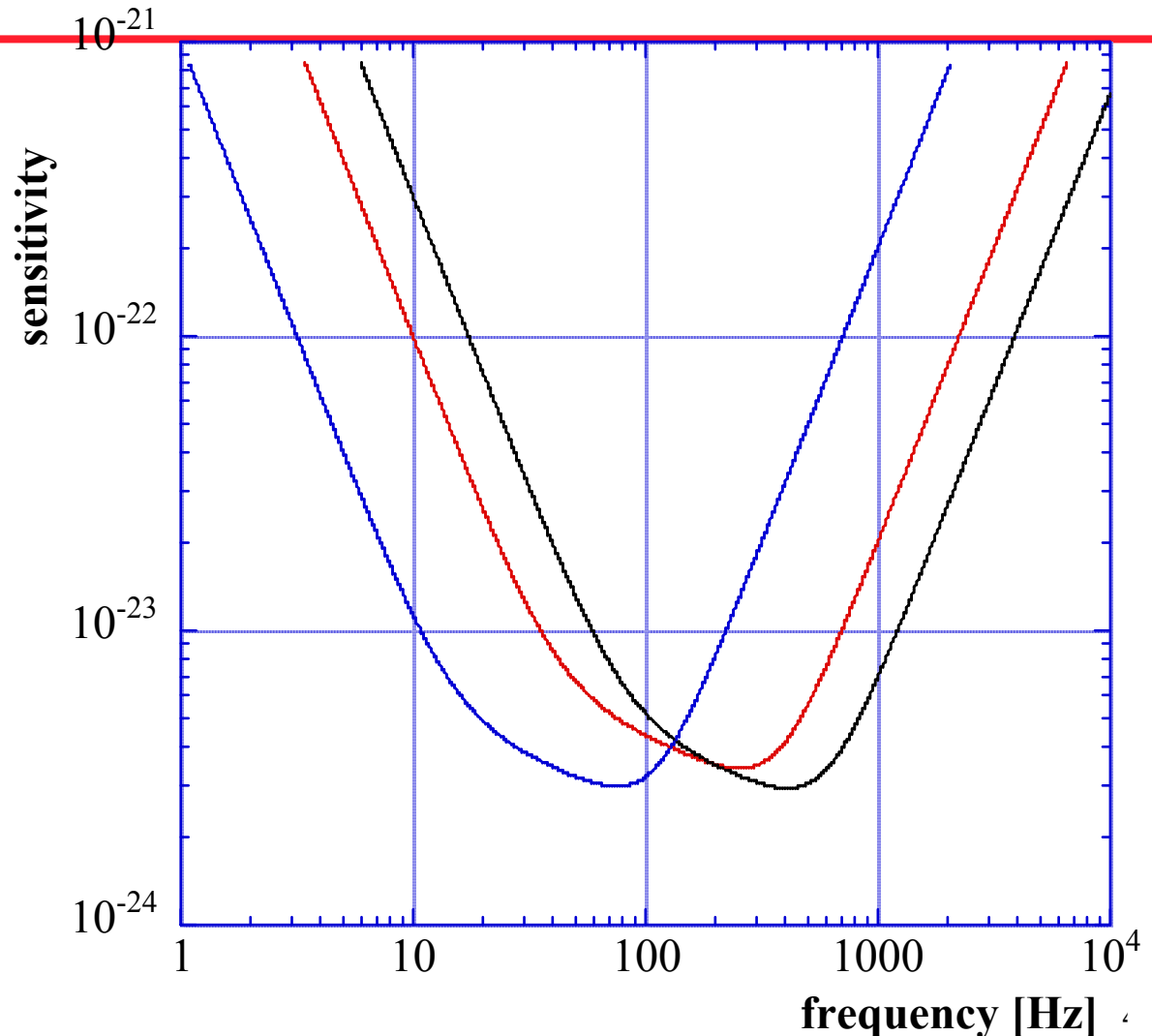


# Shifting the canyons

canyons

- advanced LIGO
- advanced LIGO \*10 power
- advanced LIGO \*.01 power 1/3 TN

To efficiently cover a large frequency span it is necessary to build dedicated Interferometers each optimized at various frequency ranges

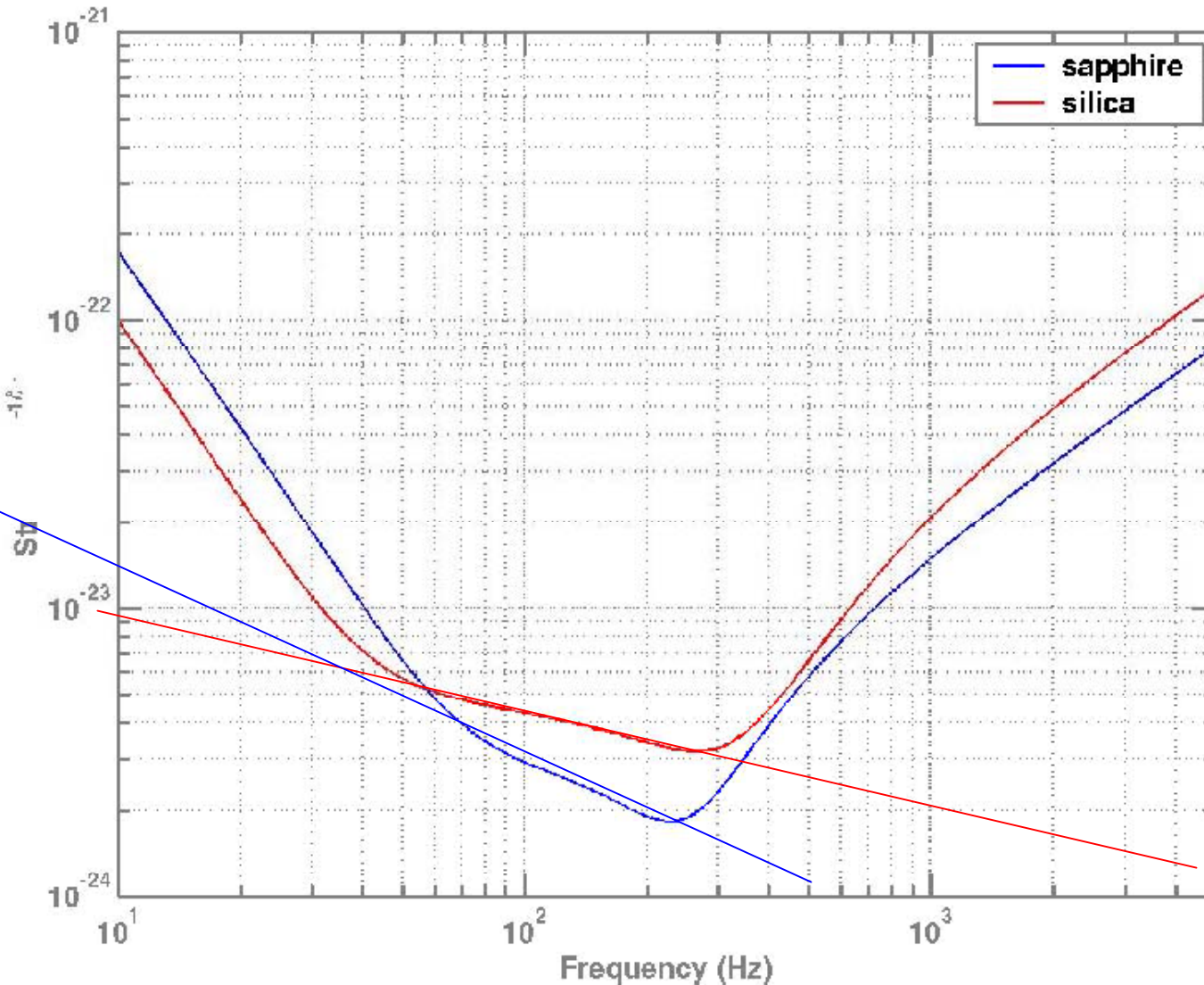


# Reasons for a Low Frequency Gravitational Wave Interferometric Detector

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- implement twin interferometers in the same vacuum enclosure
- Complementary in frequency range
- Separately cover the high and low frequency range
- Also at LF not having power limitations, fused silica is better than sapphire

# Comparing the canyons



- Different TN slope of
- Sapphire Best at high frequency Also needed to dissipate high power
- and
- Fused Silica Best at low frequency



# LIGO 4 Status of Mirror's thermal noise

## A Substrates

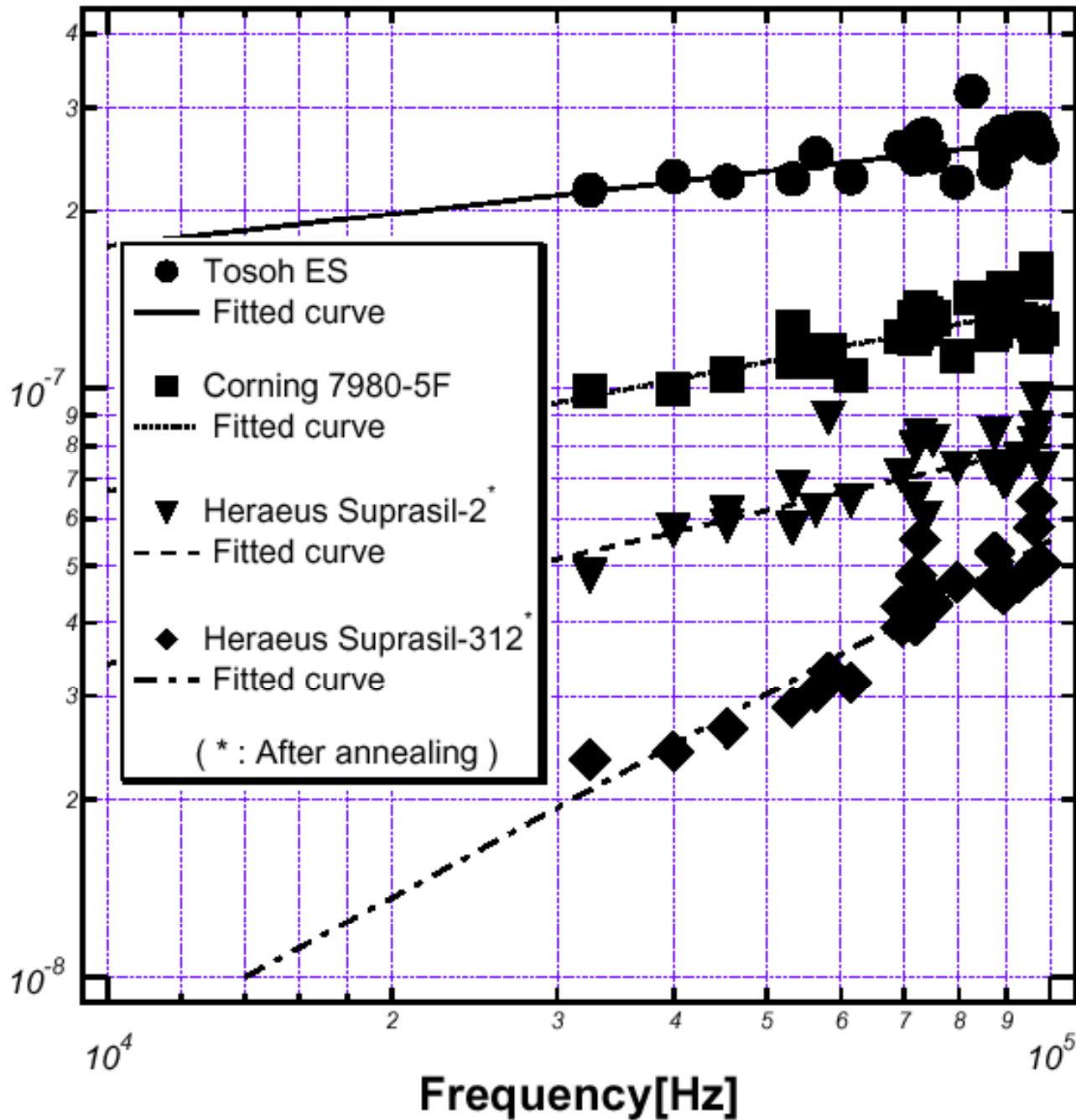
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- High internal Q factors (Kenji magnificent job!)
- May be good enough for next generation of LF-GWI already at room temperature!!!
  - At L.F. fused silica likely to be OK for long time
    - Low power densities
  - Substrates up to 75 Kg OK, more possible?



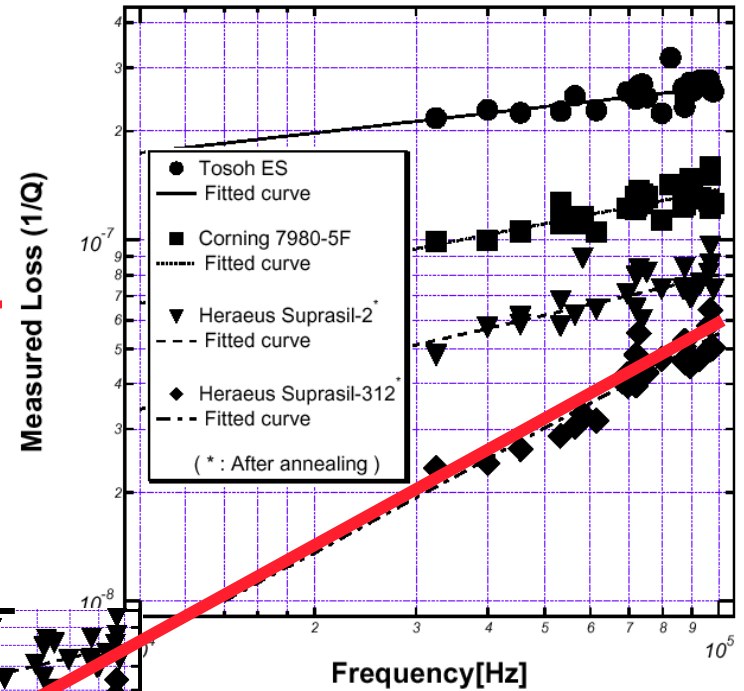
Annealing seems to expose the plunge to zero dissipation at zero frequency

Measured Loss (1/Q)



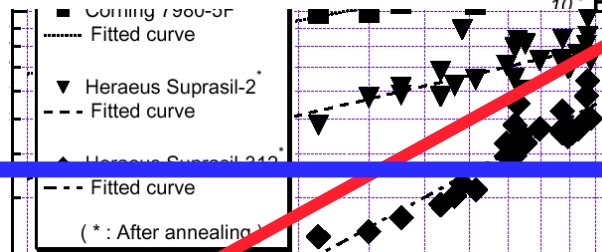


Let me **cheat** for a moment



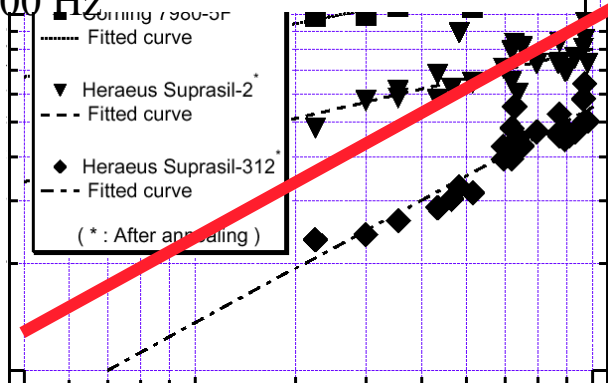
1000 Hz

Surface and  
Coating losses?



$10^{-9}$

100 Hz



$10^{-10}$

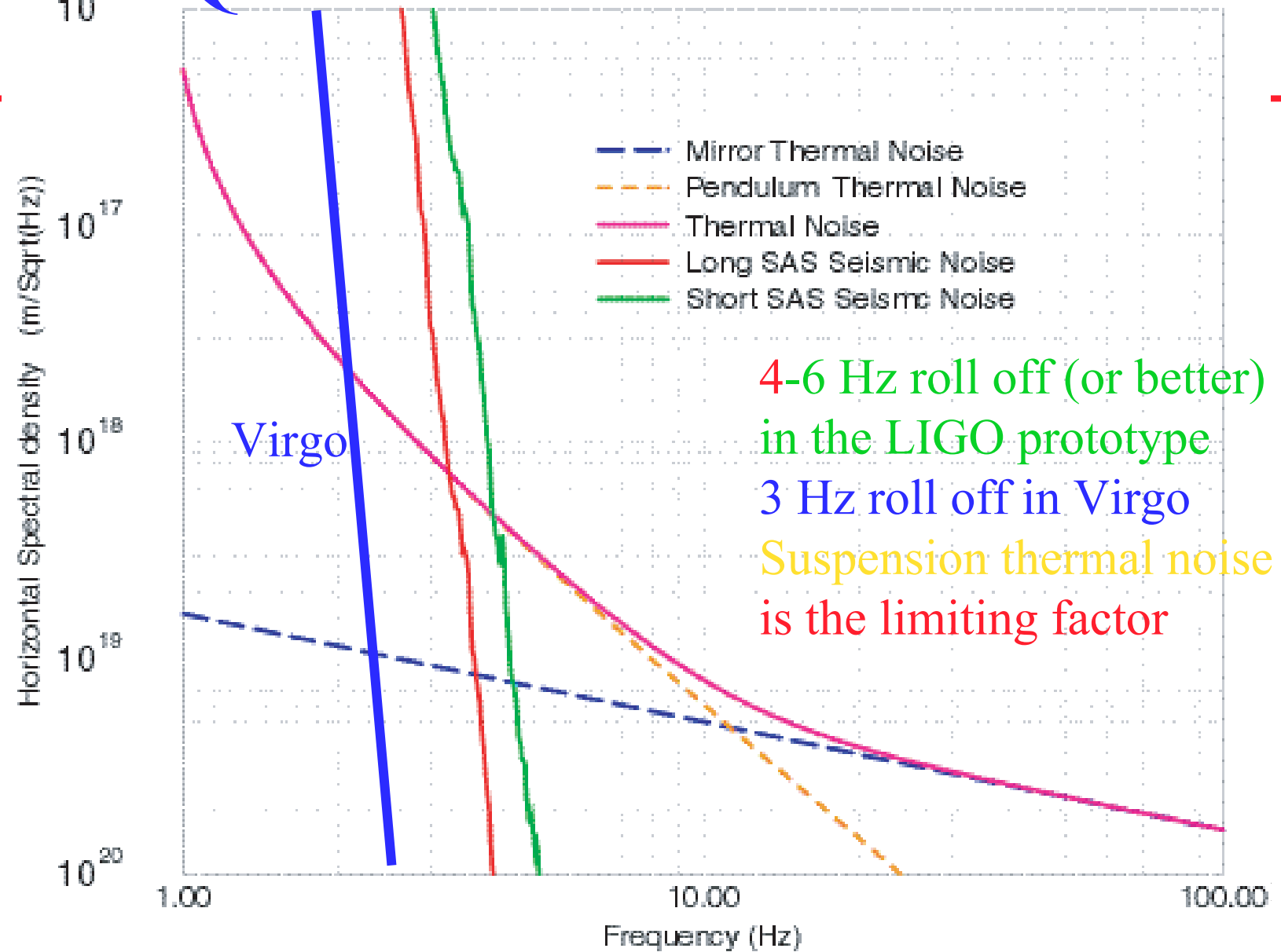
Warning: it remains to characterize  
the coating loss contributions !

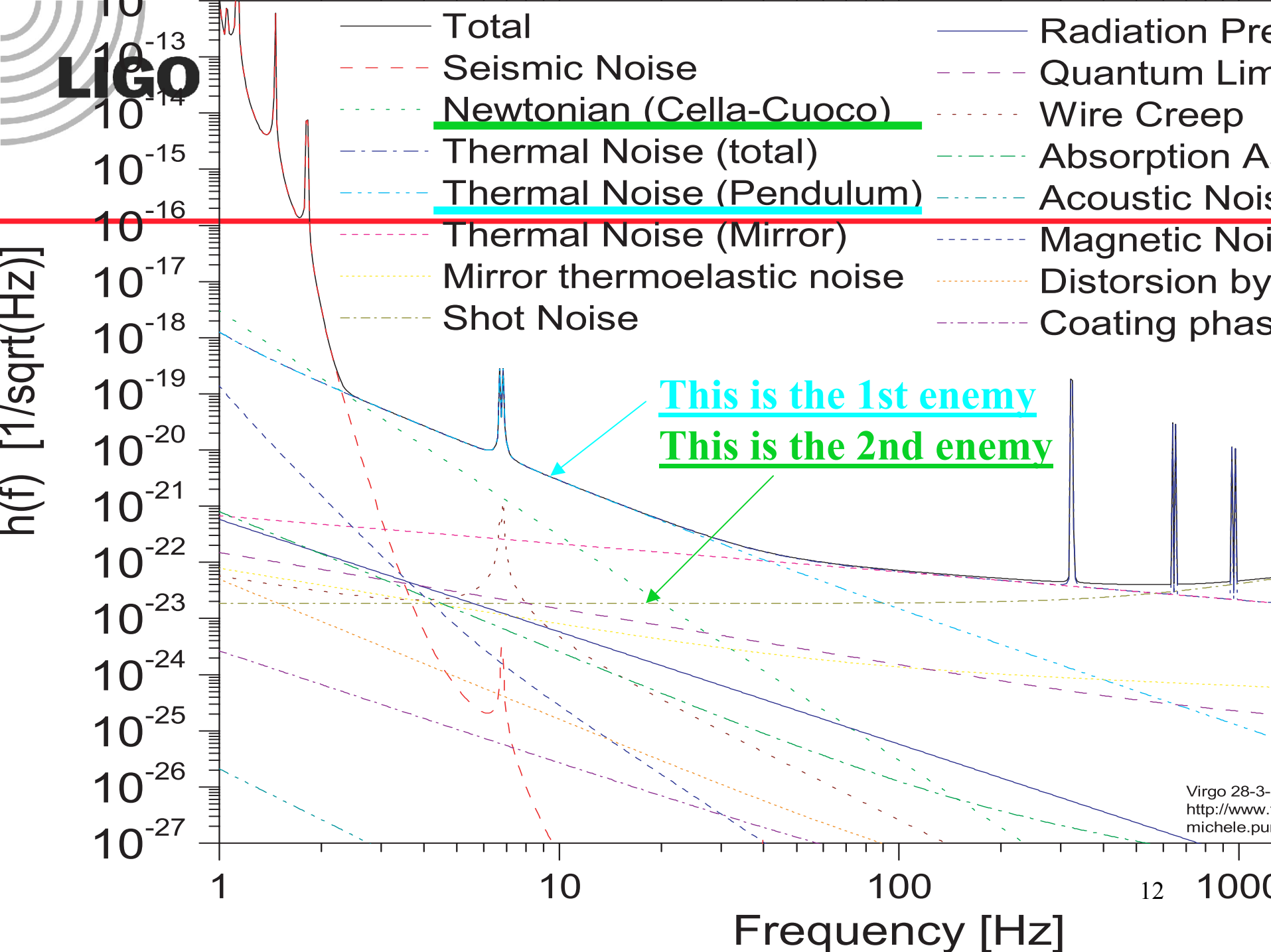
## Next priority towards a LF-GWID

- If the mirror thermal noise is OK what else is needed for a LF-GWID
- The stumbling block for a Low Frequency Gravitational Wave Interferometer is
- Suspension Thermal Noise



# Quick status of seismic attenuation





# Suspension thermal noise

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- Three avenues explored
  1. Quartz
  2. Cryogenics
    - Fibers
    - Flex joints
  3. Alternative technique
    - Glassy metals (reduced thermal noise)
    - Alternative geometries (noise at lower Freq.)

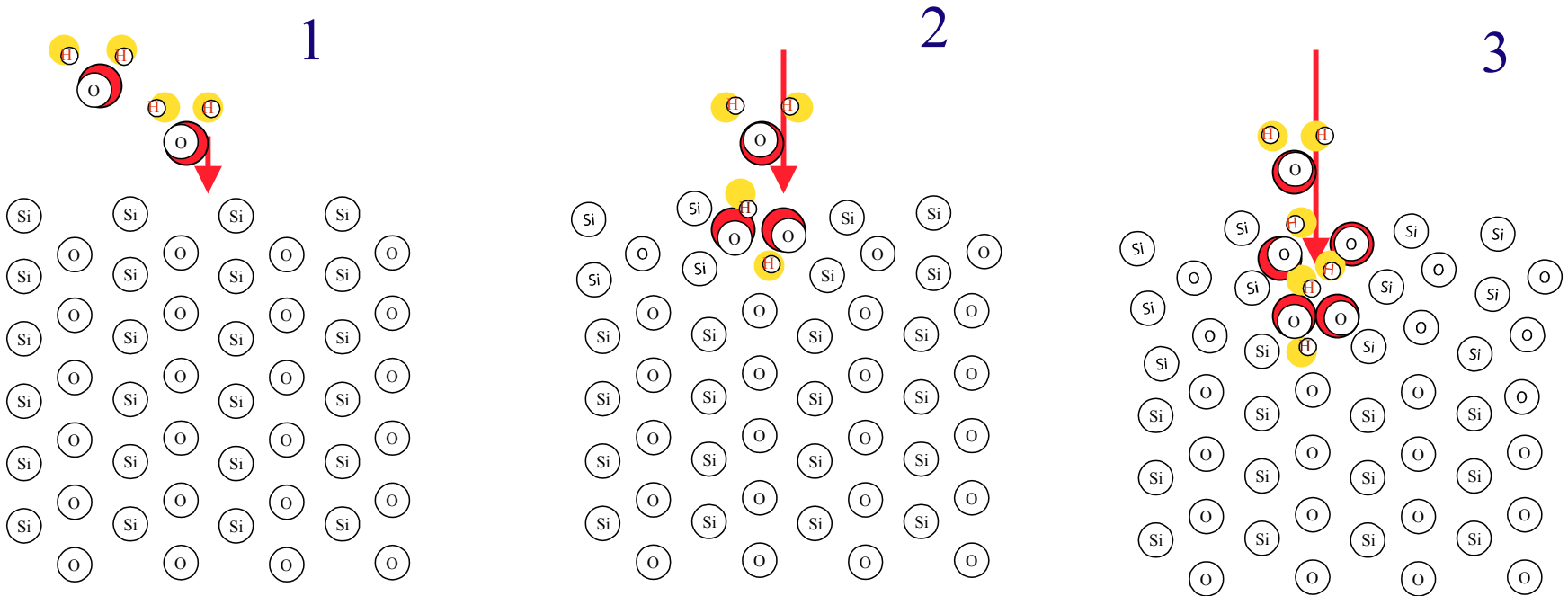
# Fused Silica Limitations effects

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- The effective fused silica Q-factor is often kept well below the theoretical  $\sim 10^8$  by surface defects
- Allowable load in fused silica fibers is limited well below one GPa by
  - reliability considerations as well as
  - Thermo-elastic noise optimization

# SiO + H<sub>2</sub>O scissor effect

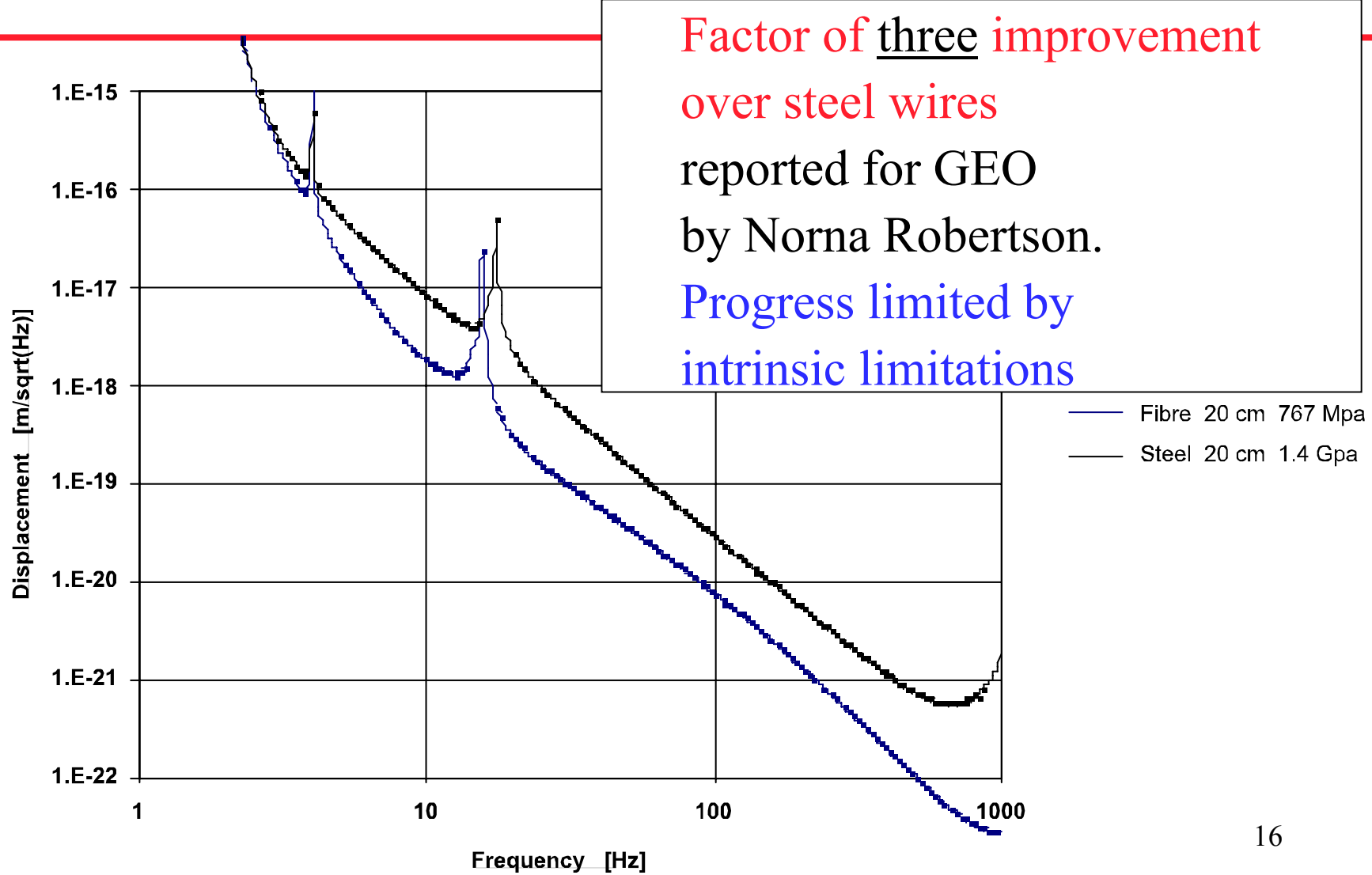
- $\text{SiO} + \text{H}_2\text{O} = 2 \text{SiOH}$
- scissor effect





# LIGO

## Triple Pendulum: Thermal Noise





# Suspension thermal noise

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- Cryogenics, a tough but in the long term almost sure bet
- Very long term project, R&D time scale imposed by power dissipation requirements.

# Suspension thermal noise

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- Used to think that cryogenics is necessary  
may be possible to deal with  
suspension thermal noise

with Glassy metals



# LIGO Alternative Suspension Solutions

## metallic flex joints

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- Metallic Flex joints have been evaluated in the past for mirror suspensions
- Extensive work by D. Blair et al.
- Flex joint have an edge because they allow large aspect ratio and large pendulum dilution factors
- Metals start **disadvantaged** with respect with glasses because of **lower** (<10,000 for metals) **intrinsic Q-factors**.

## Rekindled interest in metals

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- Better understanding of **glassy metals** recently opened **new opportunities**

# Why Glassy Metals ?

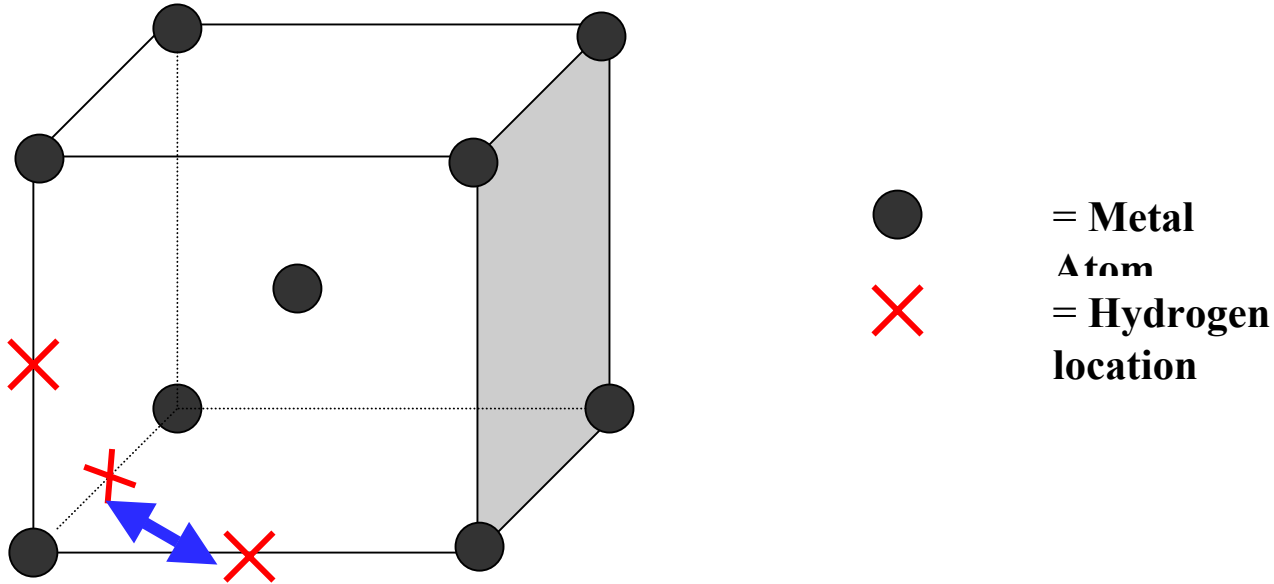
- **Easy to shape** : allow the achievement of advanced engineering and mechanical geometries.
- **Naturally** produced in  $\mu\text{m}$  thin **ribbons**.
- Not fragile
  - Critical crack length 100  $\mu\text{m}$ , few nm for Fused Silica
- Allow **loads of 4, 5 or even 6 GPa!!!**
- Best steel limit at 1.8 Gpa, typical fused silica 0.7 GPa
- Very **large elasticity** limit (2%)

## Why Glassy Metals were overlooked ?

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- Available glassy metals:
- Have large hydrogen sloshing losses leading to very low internal Q-factors
- Have been developed as braze low eutectic melting point materials
  - the mechanical Q-factor is proportional to the ratio between melting and room temperature.
- Refractory metal glasses have
  - large Q-factors
  - High melting points

# Hydrogen flipping losses



Hydrogen atom flip-flop  
with changing stresses

# Which Glassy Metals are promising

- Glassy metals can be manufactured
  - Starting from many metals, recipe:
  - Mix two close relative metal
  - Add Boron to frustrate the formation of crystalline structures
  - Cool rapidly
- Molybdenum                      mixed with  
    Ruthenium                      and additio  
    Boron



## Which Glassy Metals are promising

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- Molybdenum    Ruthenium    Boron

do not absorb hydrogen

and have

very high melting points

(similar than Fused Silica)

# Melting points

<b>Element</b>	<b>Melting Point (°C)</b>
Mo	2617
Ru	2310
B	2300
W	3410
Re	1966
Si	1410

<b>Glass</b>	<b>Melting Point (°C)</b>
MoRuB	1400-1450
WReSiB	1600-1700

do not absorb hydrogen

## Which Glassy Metals are promising

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- There is no qualitative difference between
- Crystalline Quartz / Fused Silica and
- Crystalline Metals / Glassy metals
  
- Crystallization time
  - Hours for Fused Silica
  - Seconds for Glassy Metals

# Which Glassy Metals are promising

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- Mo-Ru bond in metallic glasses and
- Si-O bond in Fused Silica,
- play same role in determining
  - the melting temperature
  - the dissipation processes and
  - the damage processes
- There is no equivalent of the OH problem
- Twining results in No fragility !!!

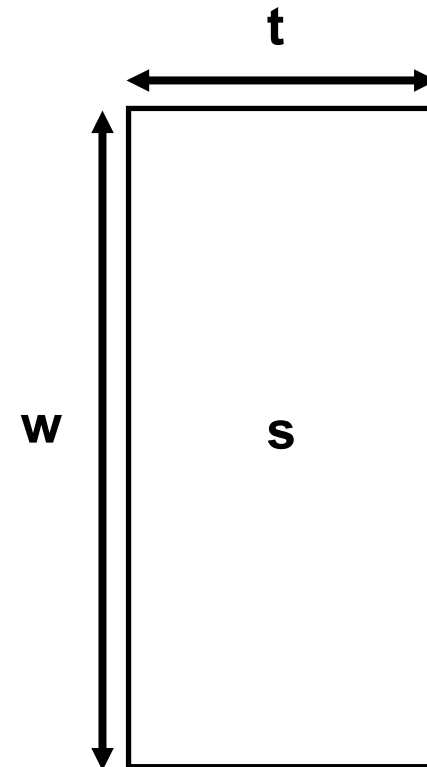
## Which Glassy Metals are promising

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- As in Fused Silica **purity and annealing** conditions will play a **determinant role**
- Unless unknown loss mechanisms are observed **MoRuB** should have high **Q-factors**
- But **intrinsic Q factor** is less important because of the **much more advantageous** possible **geometries**

# Effect of Geometry on Q Factor

- Q Factor
- $Q_p = Q_o (1 + K_G/K_{SP}) \approx Q_o (K_G/K_{SP})$
  
- Restoring Force of Spring
- $K_{SP} = EI$
  
- Inertia
- $I = wt^3/2 = st^2/2$



## Which Glassy Metals are promising

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- the glassy metals can be safely loaded to close to its limit breaking point which was measured is
- 4-5 GPa for MoRuB and
- Tough monoatomic surface oxyde and glassy structure impede corrosion and safely allow  $\mu\text{m}$  thick membranes.

# Estimated **MoRuB** glass properties

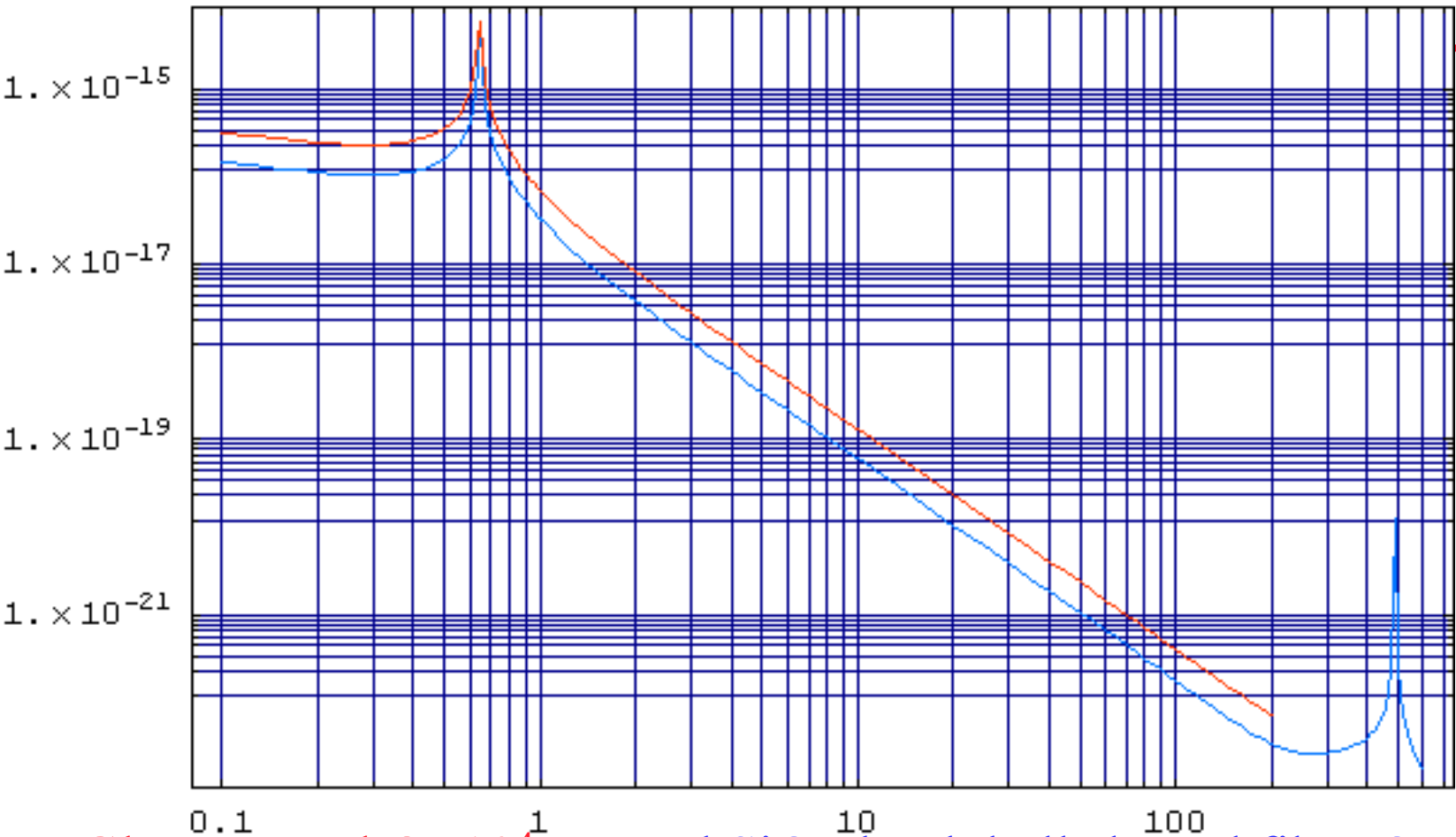
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- Mo<sub>49</sub>Ru<sub>33</sub>B<sub>18</sub> in atomic percent.
- density, 9.5 g/cc
- heat conductivity, 10 Watts/m-K
- heat capacitance, 30 J/mole-K
- linear thermal expansion coeff.,  $5-6 \times 10^{-6} \text{ (K}^{-1}\text{)}$
- elastic modulus, 250 GPa
- Poisson modulus, 0.36-0.38
- breaking point 5 GPa  
(not fragile, loadable to 4GPa) >
- - These numbers should be accurate to +/- ~20%





# Thermal noise of MoRuB flex joints



Glassy metal  $Q=10^4$ , Fused SiO<sub>2</sub> dumb bell shaped fiber  $Q=8.4 \cdot 10^8$ ,  
 10\*3000 = 30,000  $\mu\text{m}^2$ , 357  $\mu\text{m}$  diameter, 100,000  $\mu\text{m}^2$ ,  
 60 Kg mirror, 40 Kg mirror

# Estimated MoRuB glass properties

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- Q factor  $\gg 10^4$ , probably  $> 10^5$
- All of these numbers  
necessary to calculate thermal and thermo-elastic  
noise  
need to be measured soon

We are re-measuring them all

# WReBSi glass properties

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- Stronger than MoRuB
- Higher loads possible (6 GPa instead of 4 GPa)
- Higher still possible Q-Factors  
( $>10^6$  instead of  $>10^5$ )
- More difficult to manufacture

## What's being done to explore possibilities?

- Make several samples of MoRuB compositions
- Measure physical properties
  - Yield point,
  - Elastic constant
  - Poisson ratio
  - Thermal capacity
  - Thermal conductivity
  - Thermal expansion coefficient . . . . .
- Measure reed (diving board) Q-factors of samples

## What else to do to explore the possibilities?

- Demonstrate feasibility of fabrication of suspension structures
- Demonstrate feasibility of attachments to mirrors without significant loss of mirror Q-factor
- Test suspension Q-factors ( $>10^8$ ) with macroscopic mirrors

## What is being done?

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- Make several **samples of different compositions**
- **Samples are made in Caltech Metallurgy department (splat cooling)**

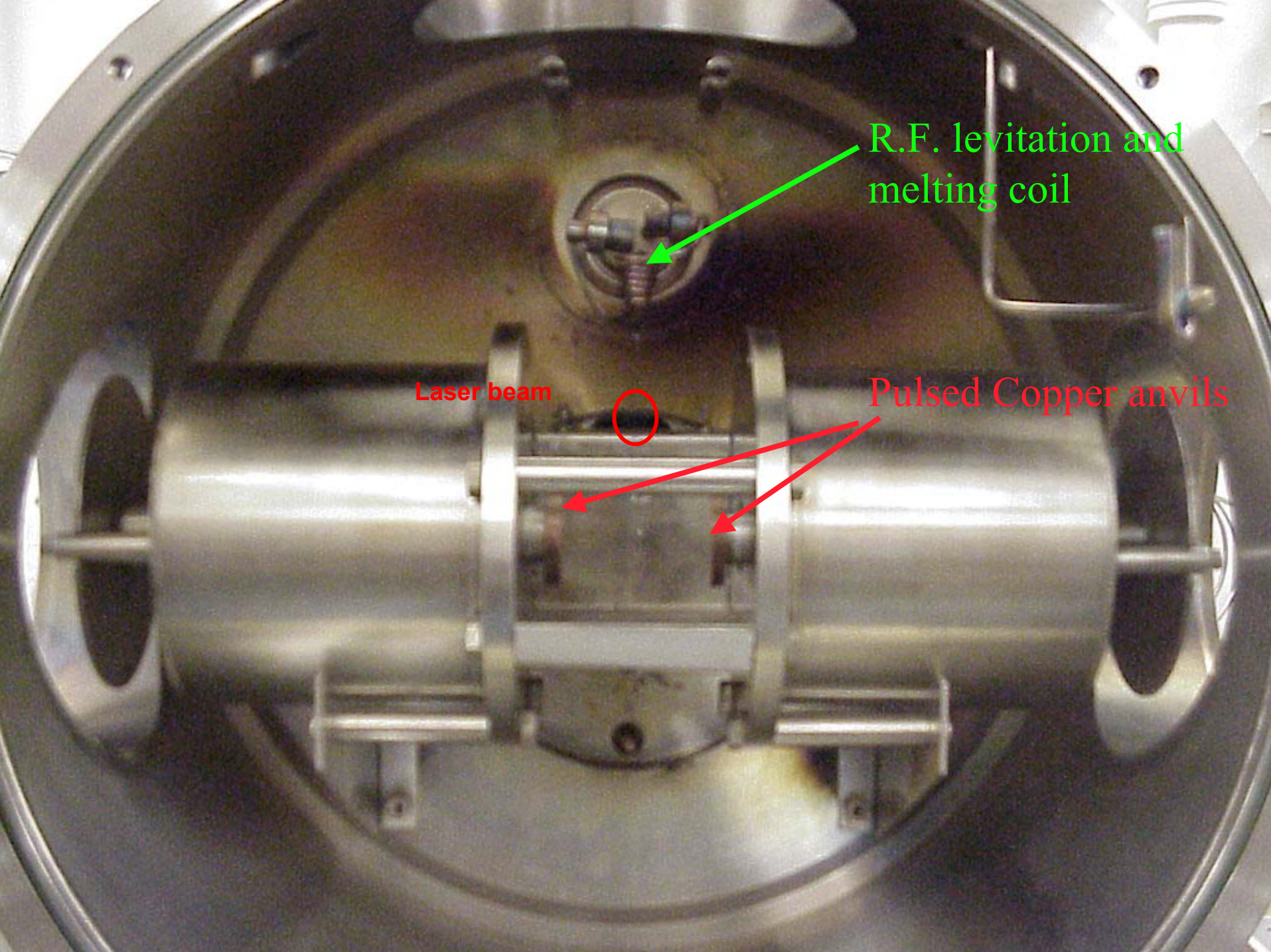
# FUSING THE ALLOY

**Mini-Arc Melter**



**Copper mould**



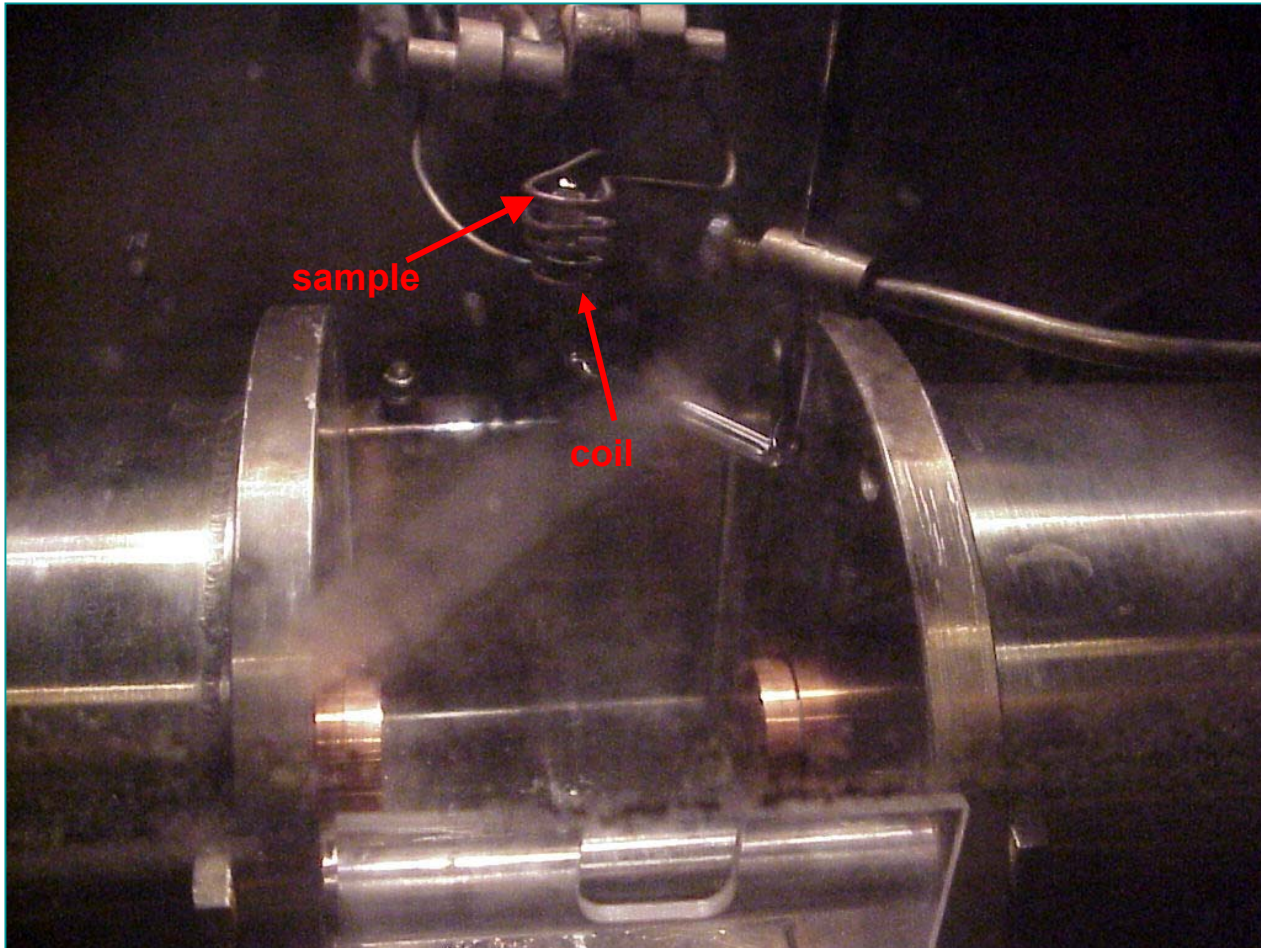


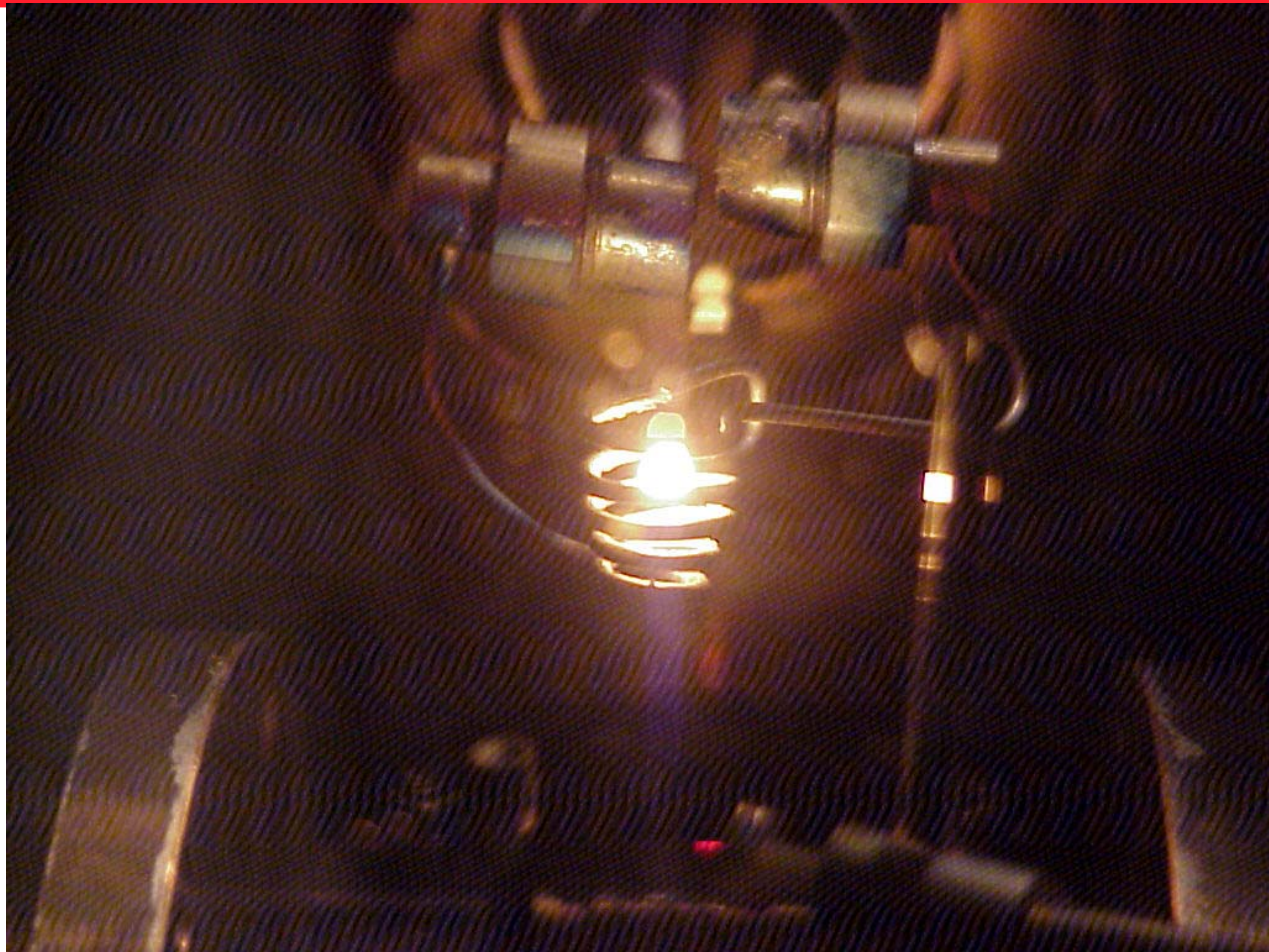
R.F. levitation and melting coil

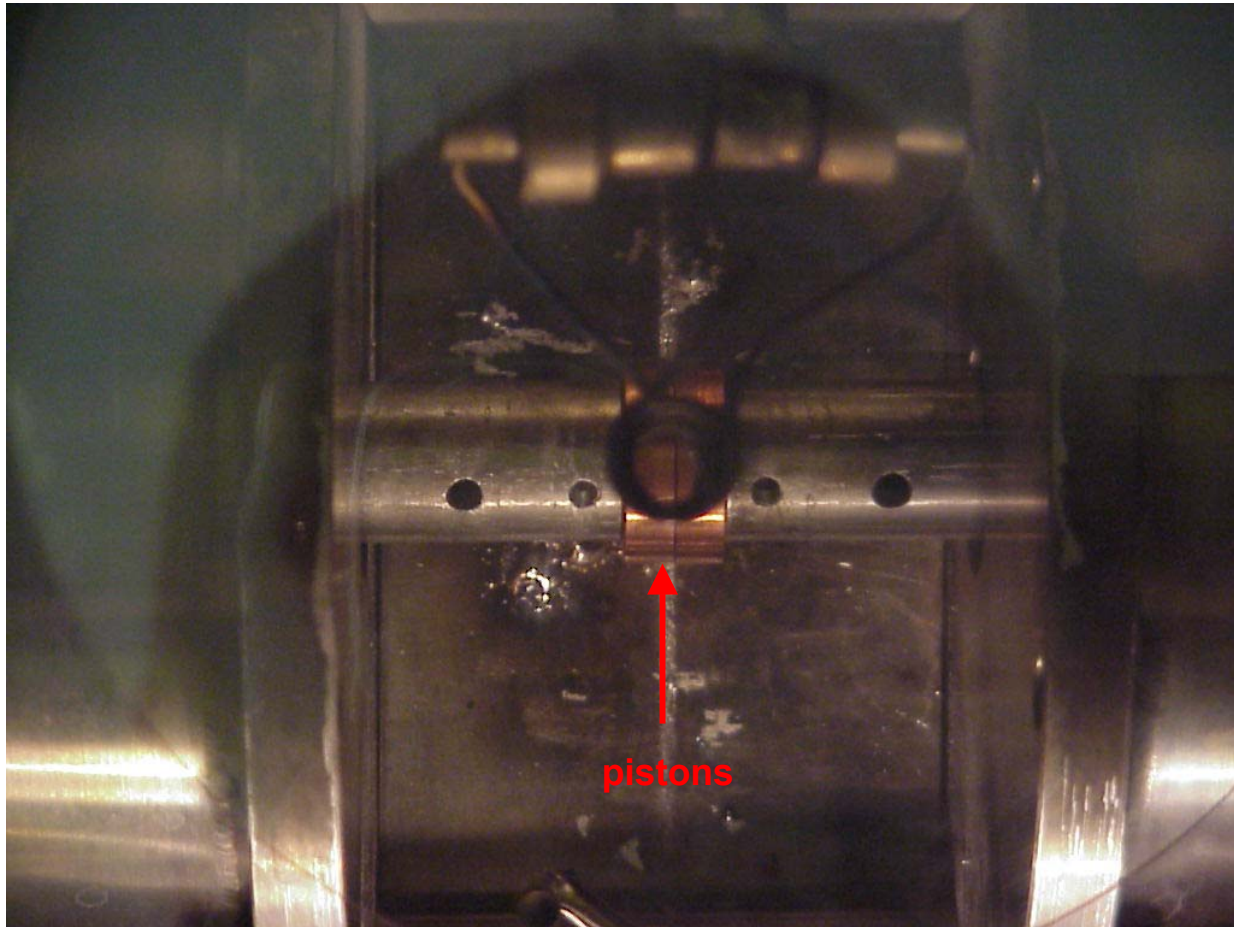
Laser beam

Pulsed Copper anvils











# What does **splat cooling** produce?

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- The end product is a **disk**
  - 50  $\mu\text{m}$  thick,
  - 15 mm in diameter
- The **surface** **copies** the anvil's surface
- Surface to be polished
- Only **3\*6 mm platelets** are required

The LIGO logo consists of several concentric, curved lines on the left side, resembling a stylized 'L' or a series of wavefronts. To the right of these lines, the word "LIGO" is written in a bold, black, sans-serif font.

**LIGO**

# Hardness and Elasticity Measurements

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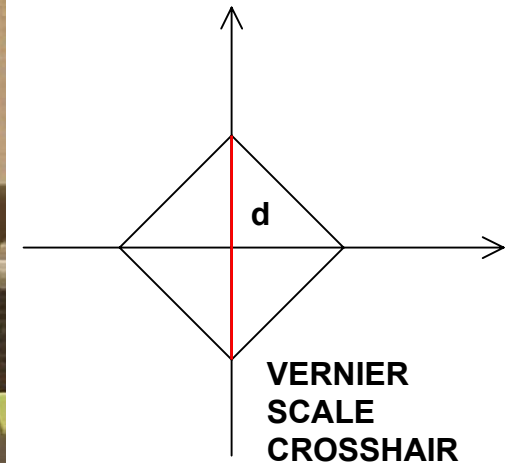
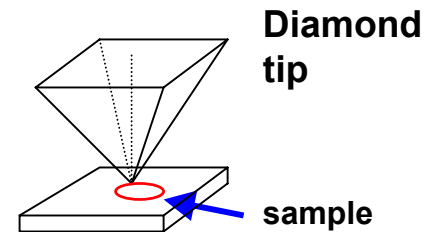
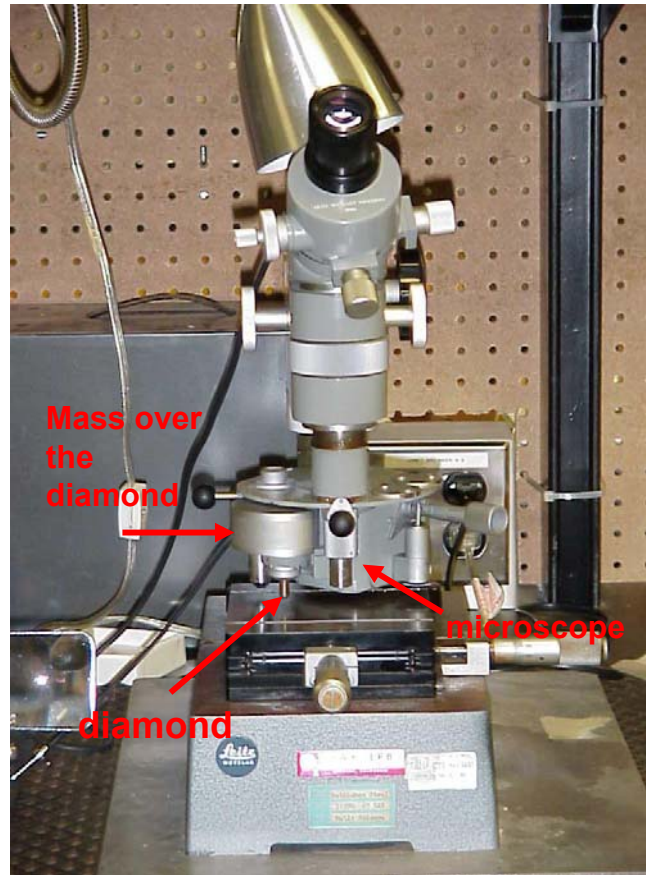
**MADDALENA MANTOVANI**



UNDERGRADUATE STUDENT  
UNIVERSITY OF PISA  
ITALY

# Apparatus For Hardness Test

- Example values  
MoRuB<sub>17</sub>  
–  $d=24.5\ \mu\text{m}$



# VICKER HARDNESS TEST

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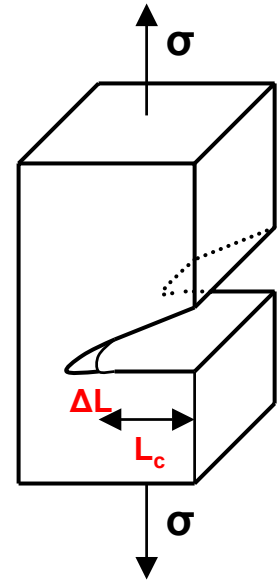
- $V.H = (1854)(m) / d^2 = [Gpa]$ 
  - $V.H.=1556.4 \text{ G Pa}$
- Tensile Yield strength:  $\sigma_Y = V.H / 300$ 
  - $\sigma_Y=5.2 \text{ G Pa}$
- 5.2 Gpa is in agreement with literature



# Elaboration

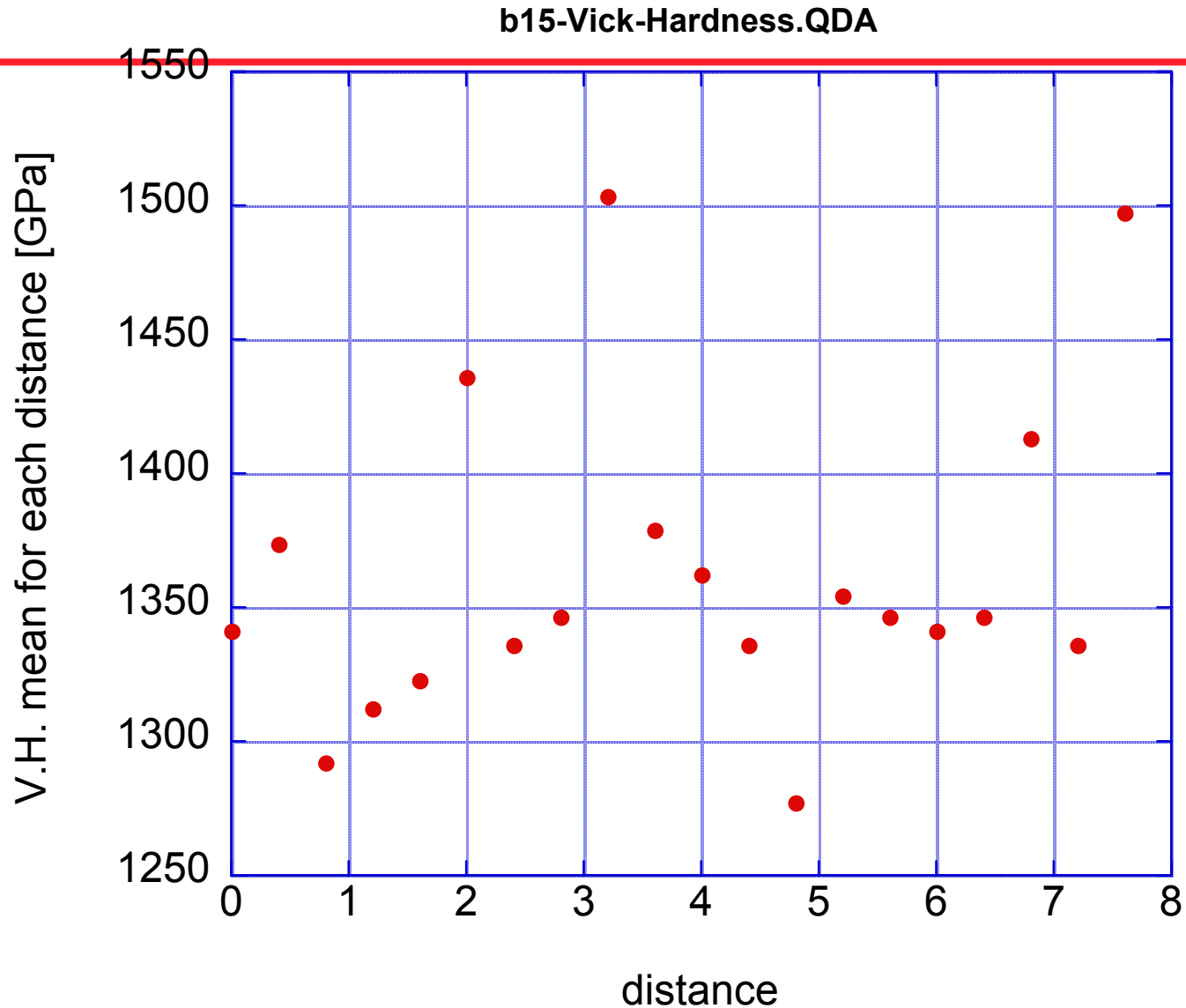
- Stress Intensity Factor/ Fracture toughness:

$$K_{IC} = 45 \text{ MPa} / \text{m}^{1/2}$$



- Critical Crack Length:  $L_c = (K_{IC} / \sigma_Y)^2$   
–  $L_c = 75 \text{ } \mu\text{m}$  (MoRuB.18)

● V.H. mean for each distance [GPa]





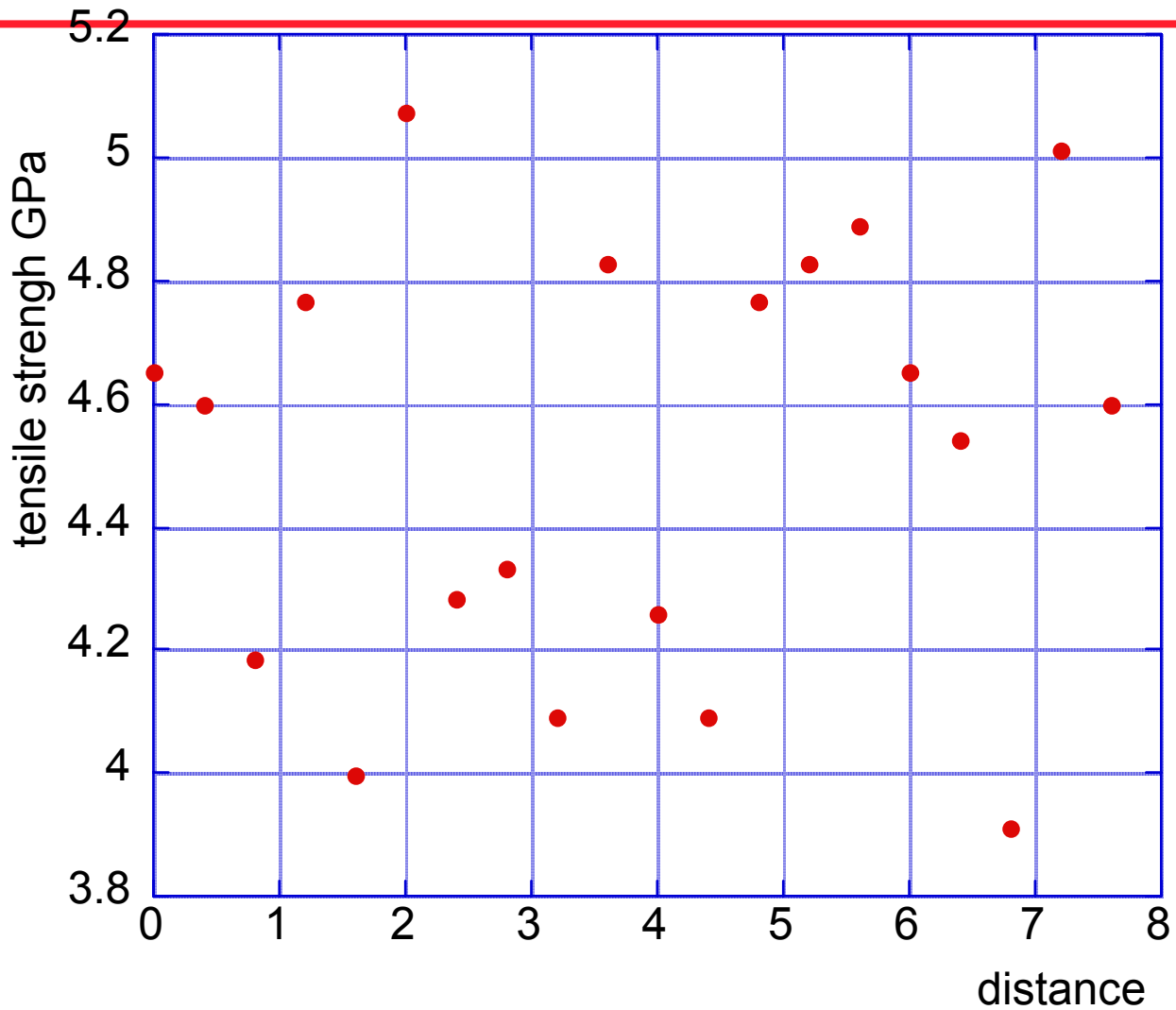
# Initial results

● tensile strength GPa

4.57+/-0.35 GPa

(5.2 for MoRuB-18)

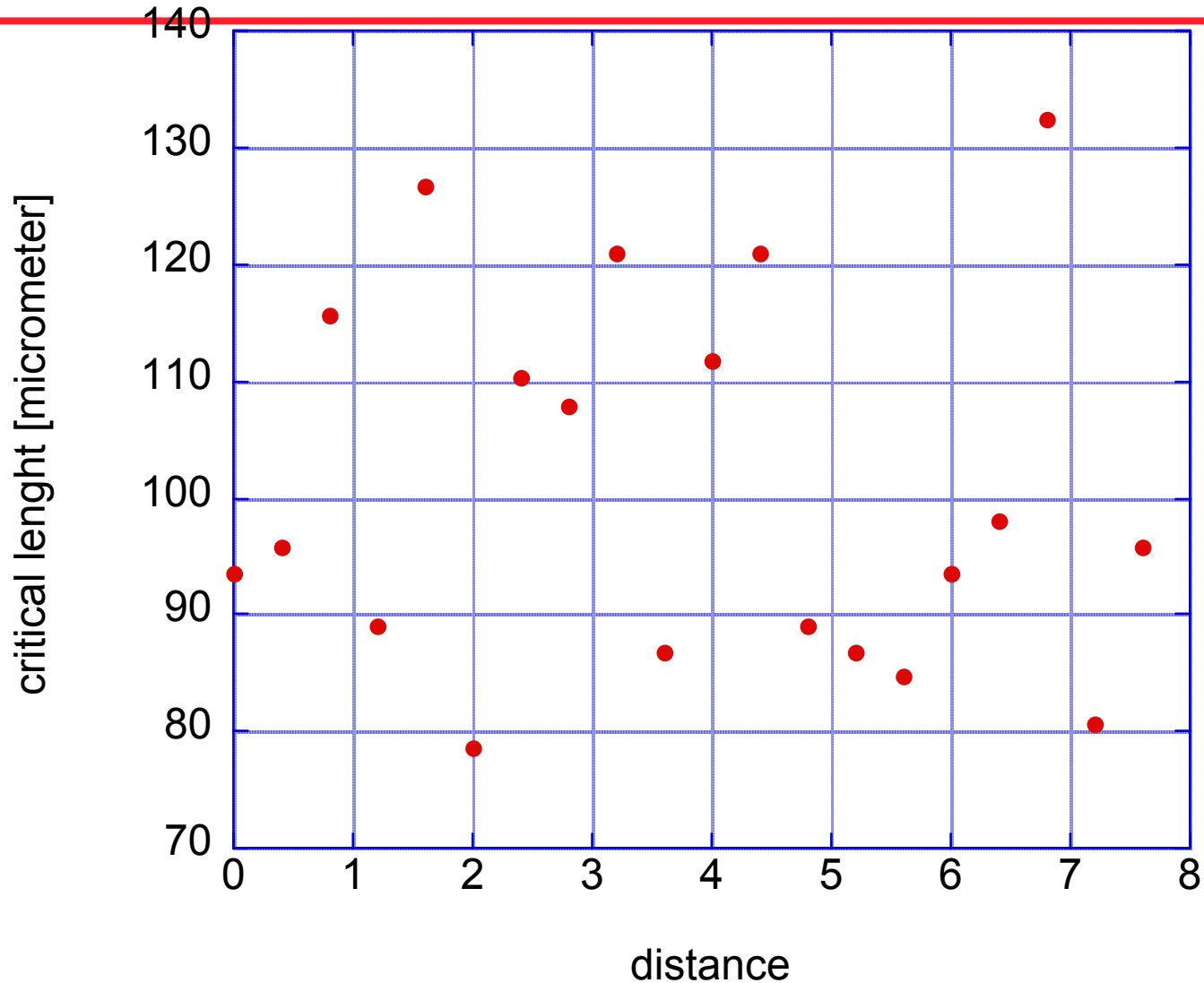
b15-Vick-Hardness.QDA



● critical lenght [micrometer]

99 +/- 15

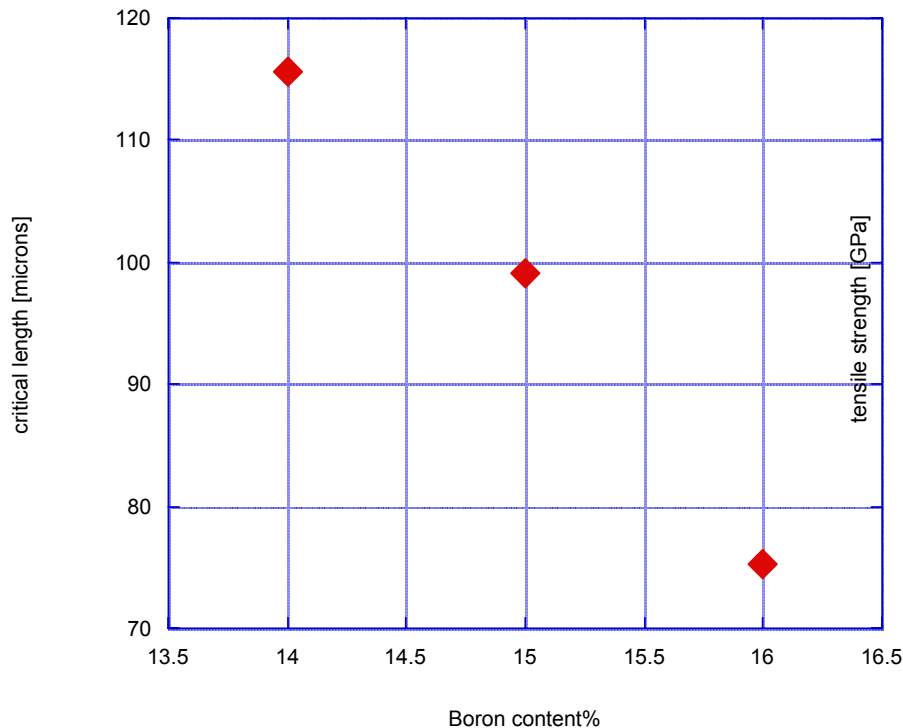
b15-Vick-Hardness.QDA



# Preliminary results hardness

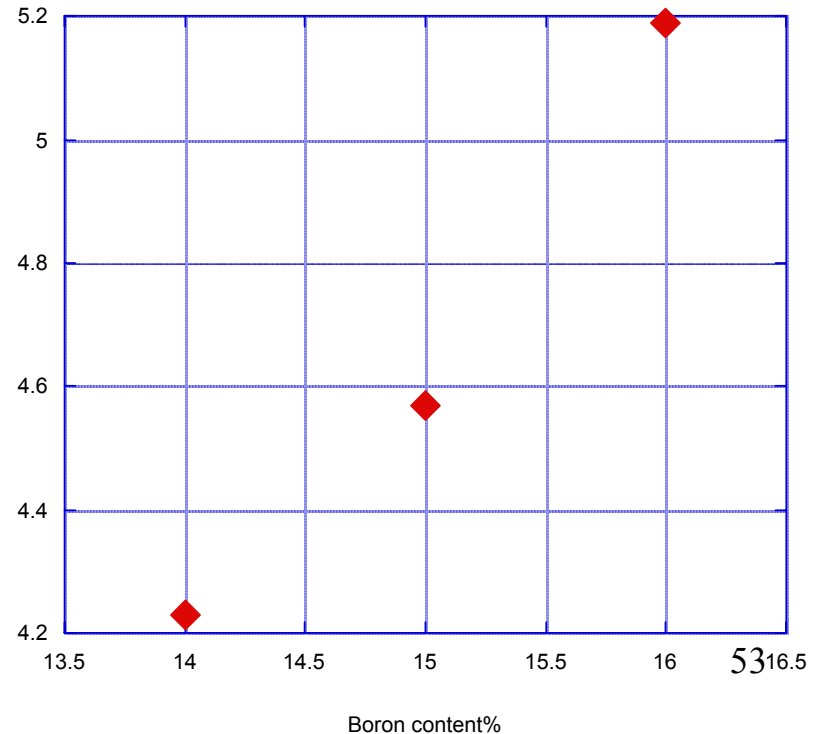
◆ critical length [microns]

tensile strength summary



◆ tensile strength [GPa]

tensile strength summary





# X-ray microdensitometry

Eric kort

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QuickTime™ and a  
Photo - JPEG decompressor  
are needed to see this picture.

QuickTime™ and a  
Photo - JPEG decompressor  
are needed to see this picture.



# X-ray microdensitometry

Eric kort

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Photo - JPEG decompressor  
are needed to see this picture.



# Differential Thermal Analyzer data

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Barbara Simoni

QuickTime™ and a  
Photo - JPEG decompressor  
are needed to see this picture.





**LIGO**

# Stress-strain behaviour of MoRuB glassy metals

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Stefano Tirelli

Undergraduate student

University of Pisa

Italy



ChenYang Wang  
Graduate Student  
Caltech

Now at University of Stanford



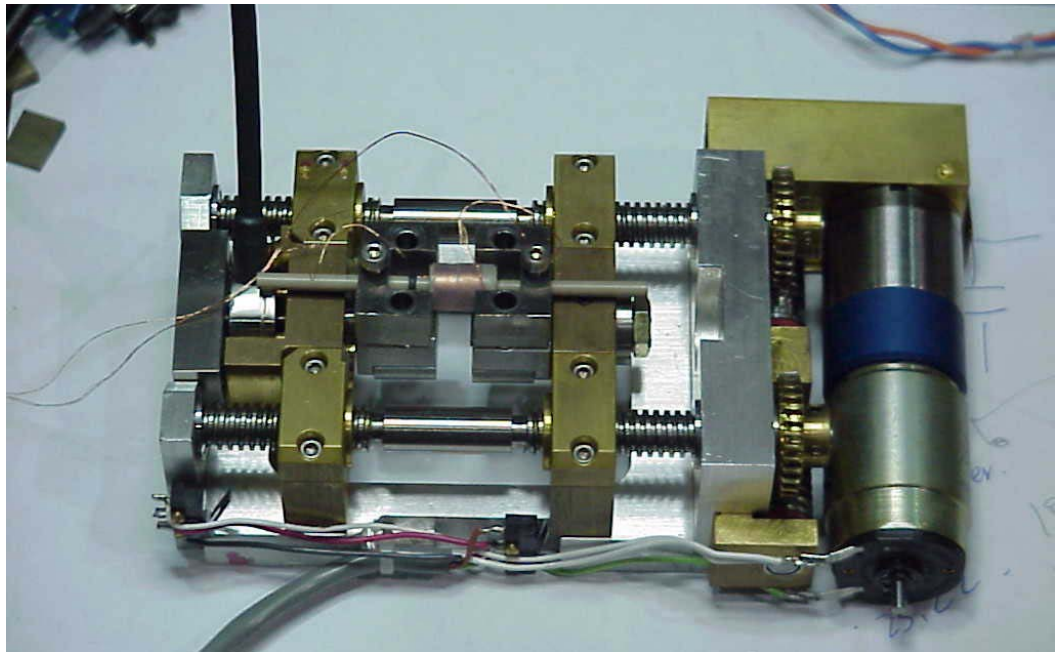
# Physical properties studied

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- Young's Modulus
  - Mechanical hysteresis
- Poisson modulus
- Yield Point
  - High yield point allows THINNER suspensions and less energy dissipation
- Structural modifications
  - Shear bands
  - Crack propagation (upcoming)



# How to study stress/strain of MoRuB?

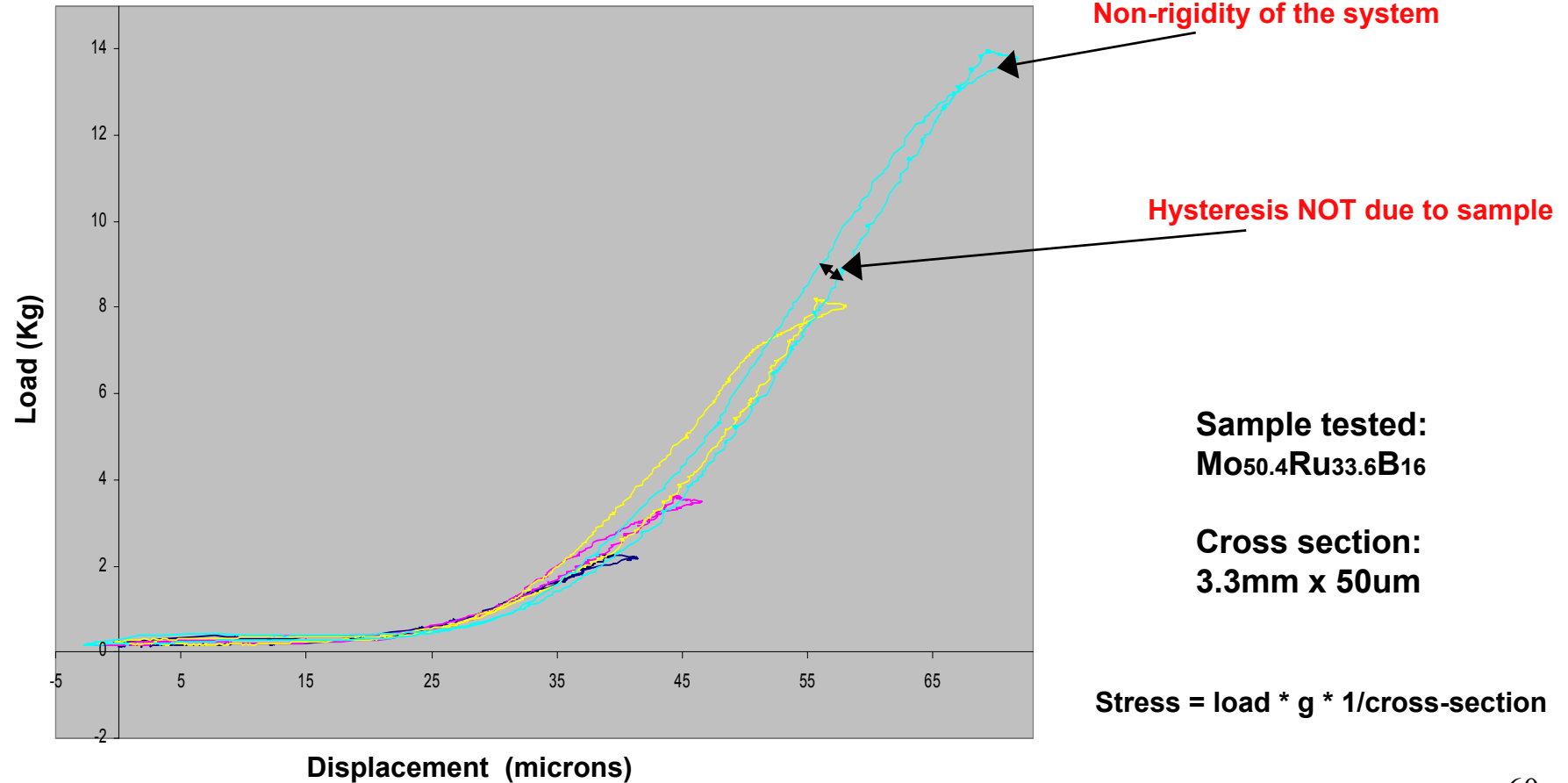


**Load frame, operational setup**

**Load frame and cell courtesy of Robert Rogan, Materials Science**

# Stress/strain chart for MoRuB

First cycles: low load



# Why low values for yield point?

**Young's modulus:**

**Boron 16: 174 GPa**

Error: ~15% mainly due to poor thickness measurement. Solution: precision-micrometer  
Upcoming.

**Yield point (lower limit!):**

**Boron 16: 1.34 Gpa**

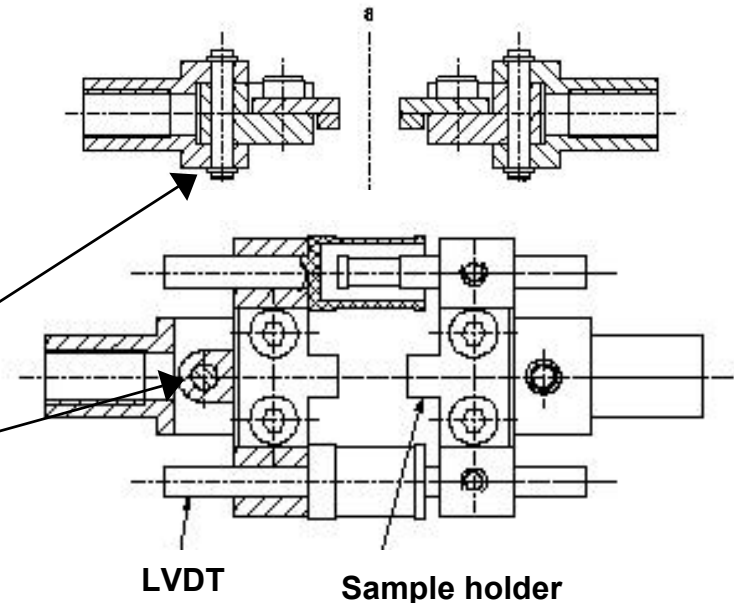
(upper limit: 5.2GPa)

**Non-uniformity of stress:** effective cross-section is LOWER than the measured one.

**Solution:** self-aligning swivel holders (already in production):

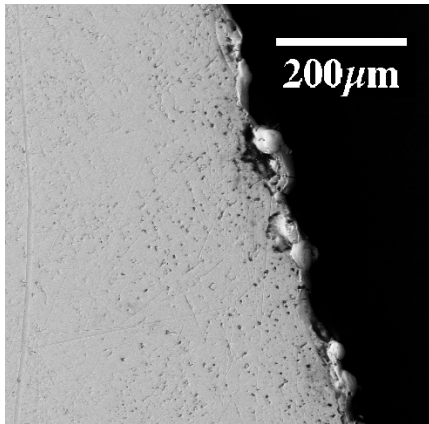
Rotating swivel

Two LVDT's for detecting torsion!

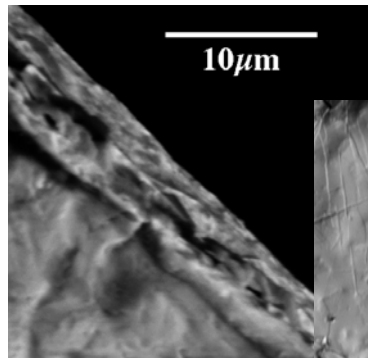


# Why low values for yield point?

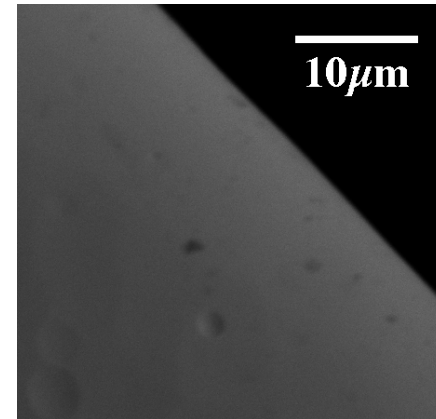
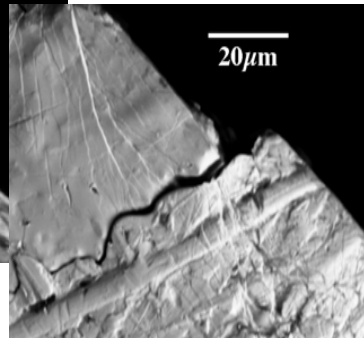
**Nucleation of cracks. To take good measurements we need regular borders without weak point for crack nucleation:**



**EDM Cut**  
Local melting  
Possible formation  
Of crystals on edges



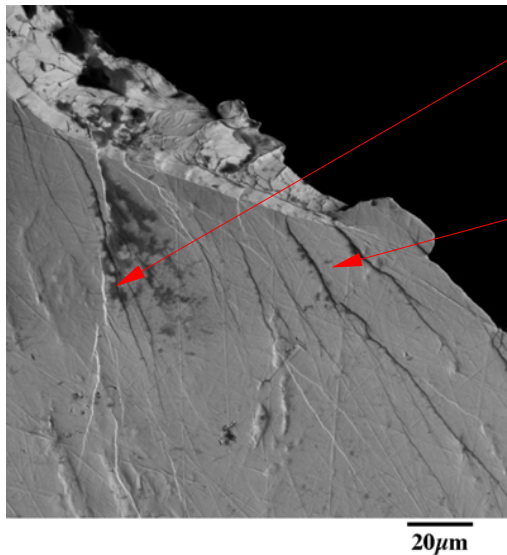
**Scissor cut:**  
Very irregular and unreliable!



**Electropolished cut:**  
The best!

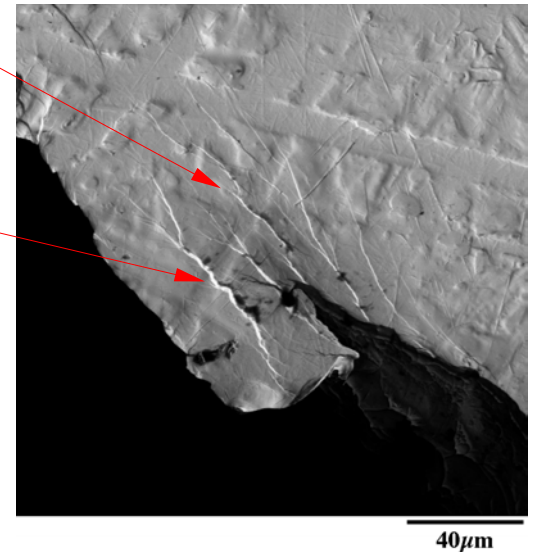
**Nucleation of cracks causes premature failure of the material!**

# Structural effects of stress: shear bands



**Shear bands**

Details of a MoRuB broken sample



**These shear bands prove that, unlike fused silica, the cracking process is not catastrophic**

**The material is resilient**

# What needs to be done:

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- **Testing on electropolished samples to obtain a value close to the one calculated by Vicker hardness test (5.2GPa).**
- **Study of crack propagation.**
- **Poisson's modulus.**
- **Observation of shear bands during formations: load frame is designed to fit into an SEM casing.**



# X-ray scattering measurements of crystallite contamination in glassy metals

Brian Emmerson

Undergraduate

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Gonville & Caius College



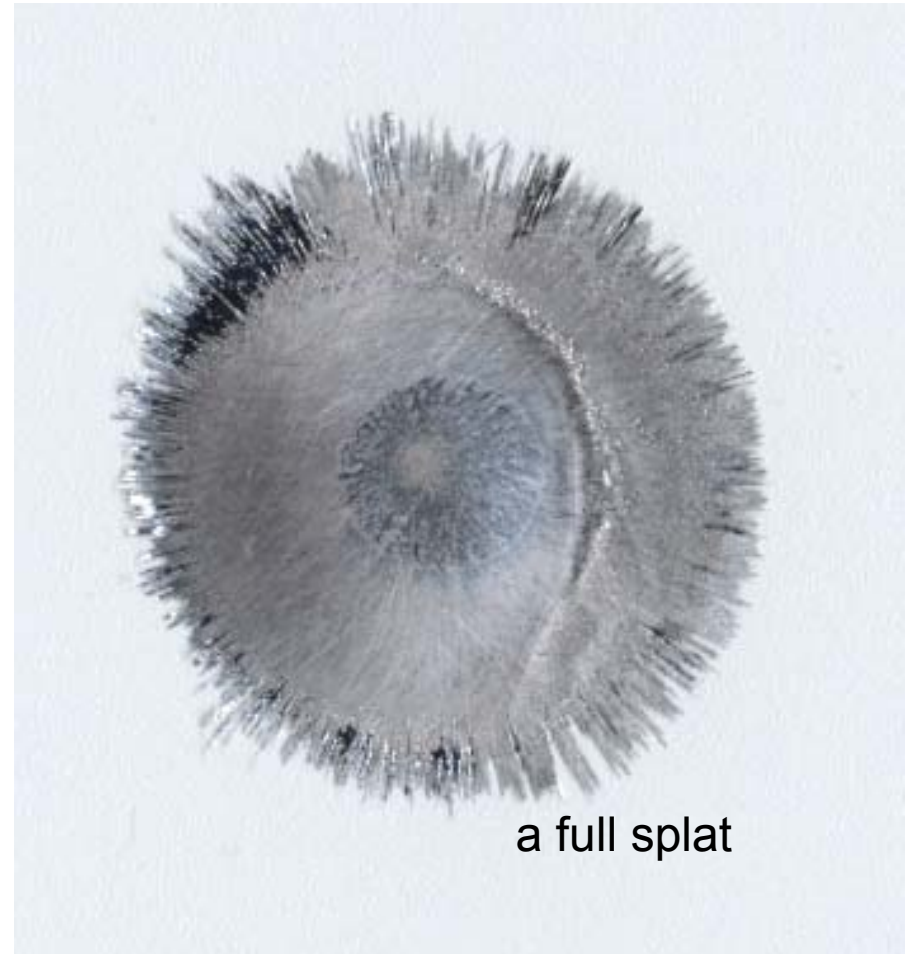
# Overview

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- Motivation
  - **why bother?**
- Theory of x-ray diffraction
  - **how it works and what it tells us**
- Results
  - **variations in samples and using diffraction to show them**
- Future options and analysis
  - **what else to do with the data?**

# Motivation: glassy metal production

- the splats aren't uniform:
  - parts of them are **glassy**;
  - (this is a **Good** Thing)
  - parts of them are partially or entirely **crystalline**
  - (this is a **Bad** Thing)



# Motivation: *Bad* crystals?

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- Crystalline-glassy interfaces within a sample are a source of local weakness
- We want to place an **upper limit** on the amount of crystalline material in a given area of sample
  - this determines how suitable that region is for use in a suspension

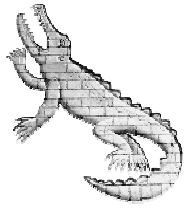
HOW?

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USE X-RAY  
DIFFRACTION

(what else?)

# X-ray diffraction theory: who and how?

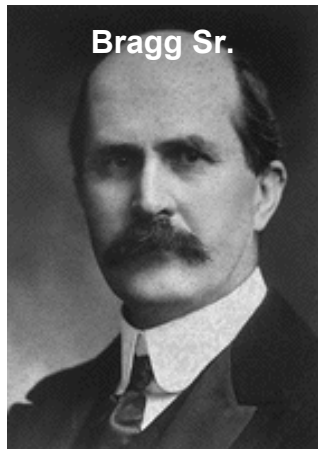


- The ‘who’ part:
  - Braggs, Sr. & Jr.

$$\text{“}2d.\sin\theta=n\lambda\text{”}$$

(Bragg’s Law)

The condition for **constructive** interference of x-rays diffracted from atomic planes in a periodic lattice, with real-space separation  $d$



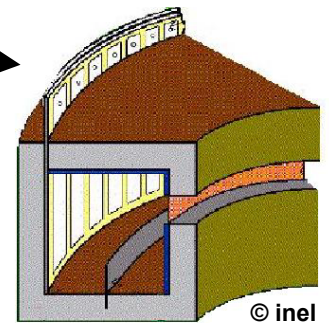
Bragg Sr.



Bragg Jr.

# Realising the technique

- Samples are irradiated with **Co K $\alpha$  radiation**, which penetrates to depths up to  $\sim 1\mu\text{m}$  before being diffracted
  - thanks to Brent Fultz and Jason Graetz for allowing us to use the x-ray equipment;
- X-rays are detected at one of 4096 channels curved “**gaseous blade detector**”
  - channels are converted to the conventional  $2\theta$  angle using a standard sample as reference;



# X-ray diffraction theory: how does it help?

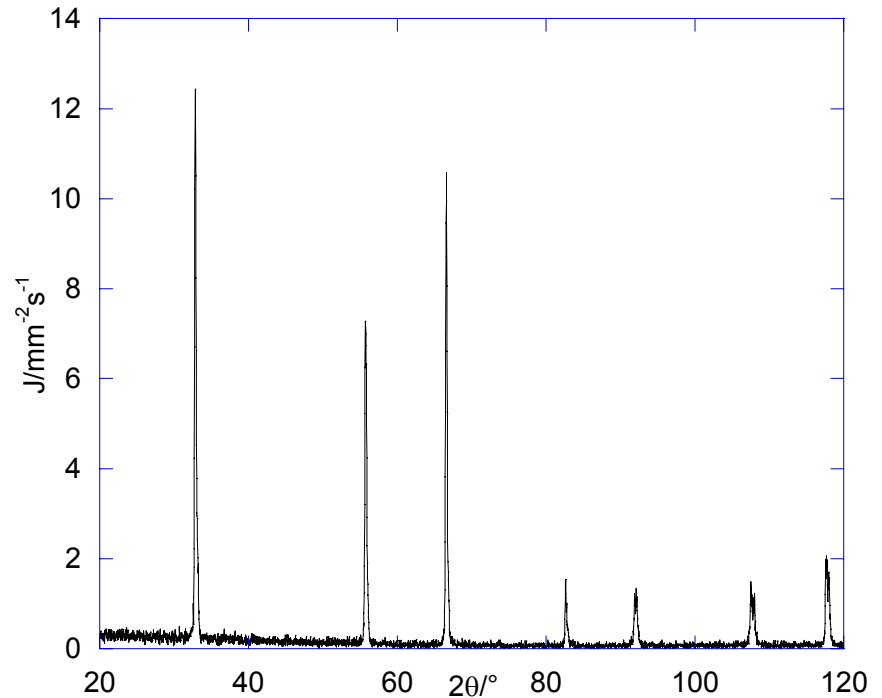
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- Peaks in a real material's diffraction pattern are never perfect  $\delta$ -functions
- The widths of peaks can be used to infer a material's composition properties
- In the diffraction pattern:
  - **glassy peaks are broad and smooth;**
  - **crystalline peaks are narrow and sharp**
  - why?



# X-ray diffraction theory: peaks for crystals...

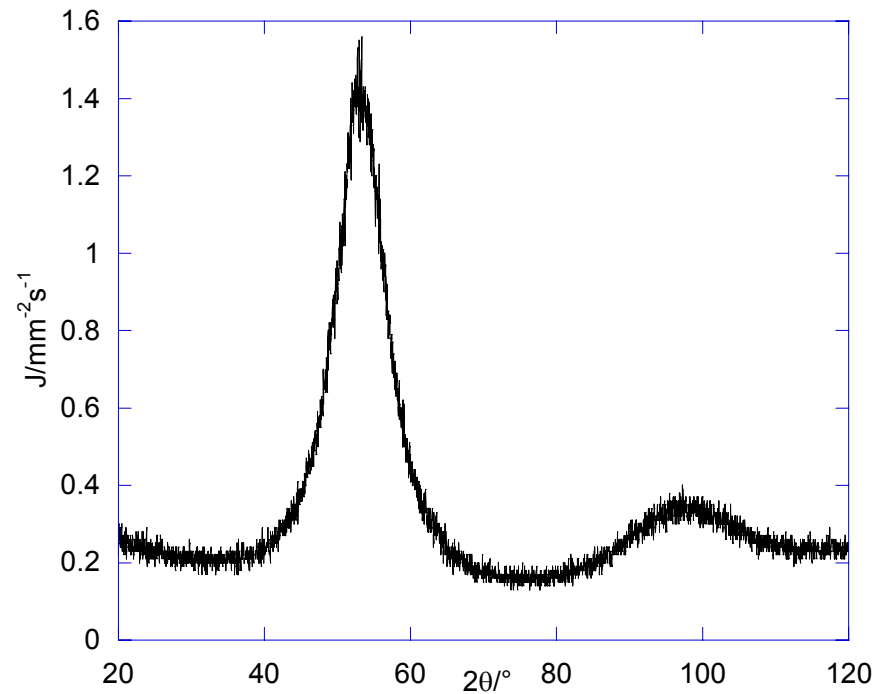
- For crystalline materials, interatomic distances are well-defined, with comparatively little spread
- This produces sharp, narrow peaks that “stick out” of the diffraction pattern – even for a mixed composition



X-ray diffraction pattern from  
polycrystalline silicon

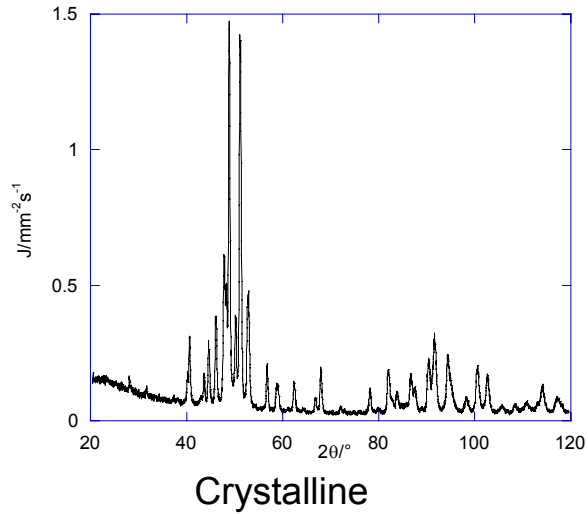
# ...and peaks for glasses

- Angular dilation of a glassy peak is chiefly caused by the distribution of atomic separations
- Constructive interference is possible over a much wider range of angles for a glass
  - this is no longer a simple Bragg relationship: it's more complicated

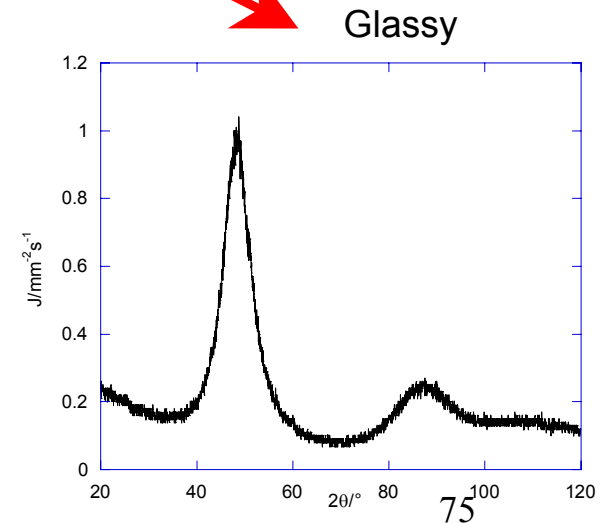
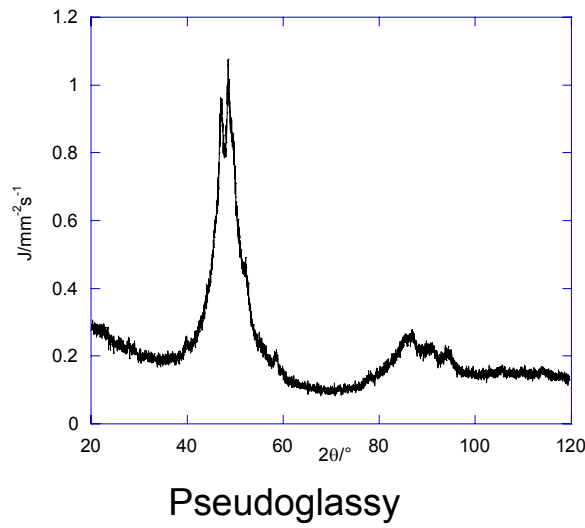


X-ray diffraction pattern produced by the homogeneous, industrially produced glassy metal, Metglas 2826MB ( $\text{Fe}_{40}\text{Ni}_{38}\text{Mo}_4\text{B}_{18}$ )

# Characteristic sample patterns



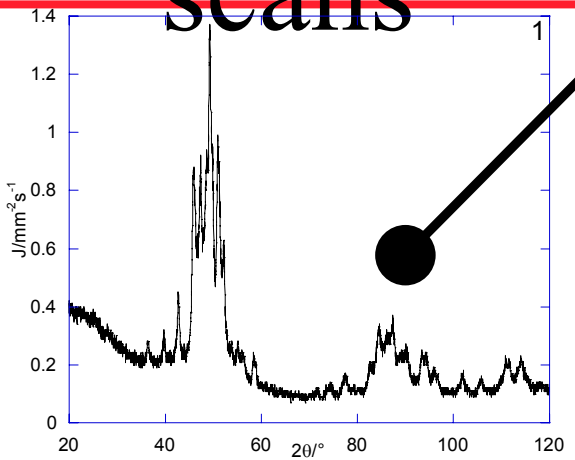
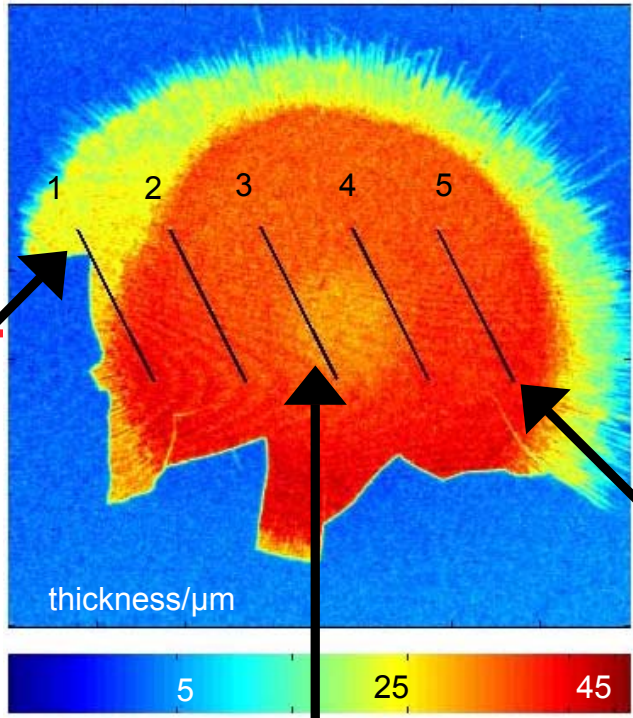
decreasing crystallite content  
for  $(Mo_{0.6}Ru_{0.4})_{83}B_{17}$



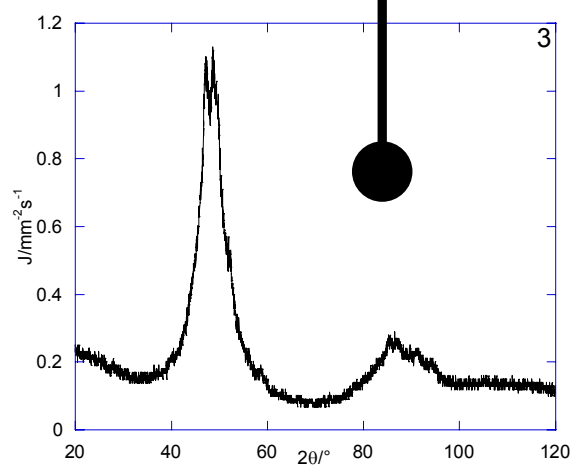


# LIGO Splat scans

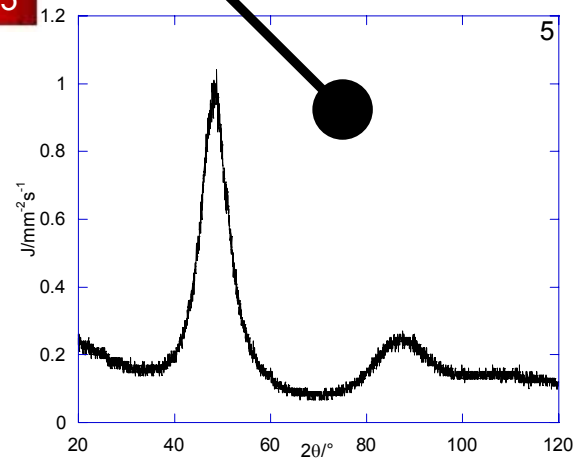
[Thanks to Eric Kort and Stoyan Nikolov for this image]



•mainly crystalline



•partly crystalline



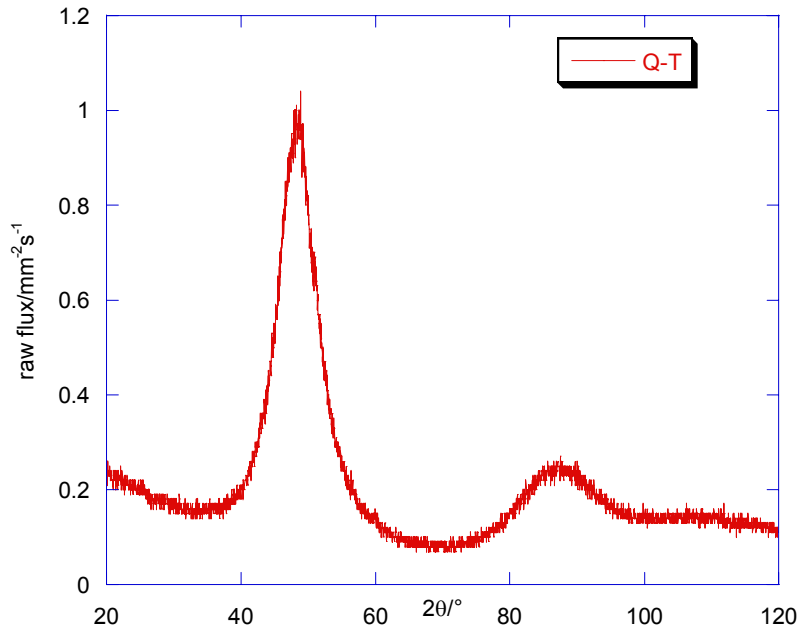
•fully (?) amorphous

# Potential for quantitative analysis

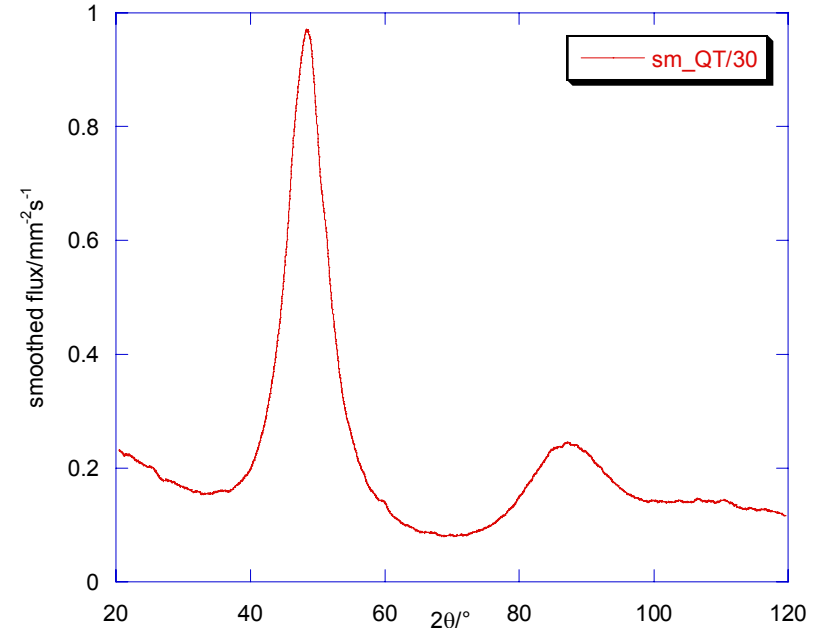
---

- At the moment, glassy regions are located by “eyeballing” diffraction data
- Can quantitative composition estimates be obtained?
- analyse data with various types of curve fitting
  - areas under peaks estimate % composition

# X-ray scattering analysis



**Start off with diffraction pattern of a very glassy data set (this is section 5 from the splat)...**



**...and smooth it out by averaging each data point with the surrounding 30 data points**

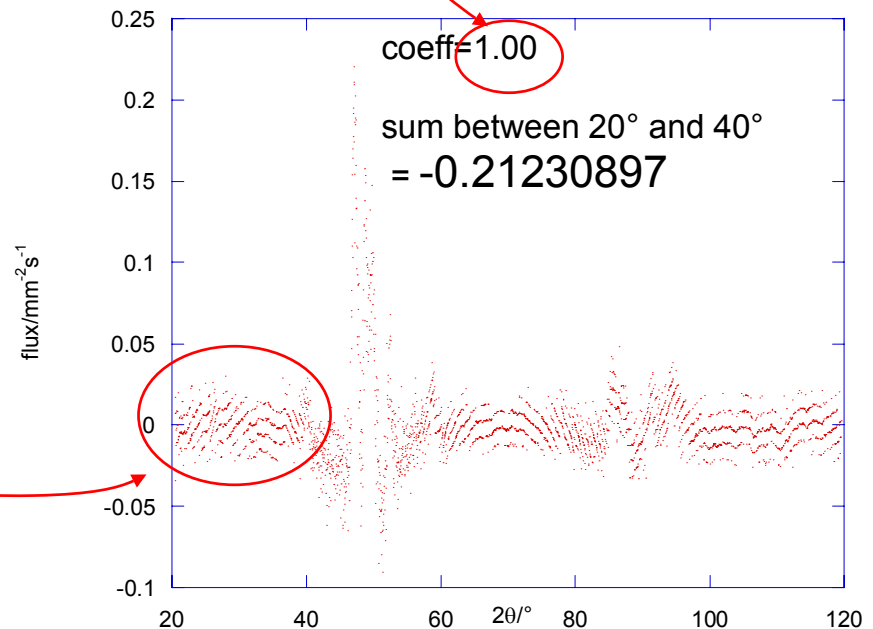
# Data “fiddling”

---

- Sum rules demand that equal exposures on chemically equal samples give same integrated flux
- Variations in detector sensitivity and x-ray intensity mean that the integrated fluxes are not always the same
  - rescale the areas of the plots to make them equal to that of the smoothed standard
- Use this rescaled data to estimate the **crystallite content**:
- Subtract a proportion of the smoothed, glassy curve from the data of interest

# How to determine this proportion?

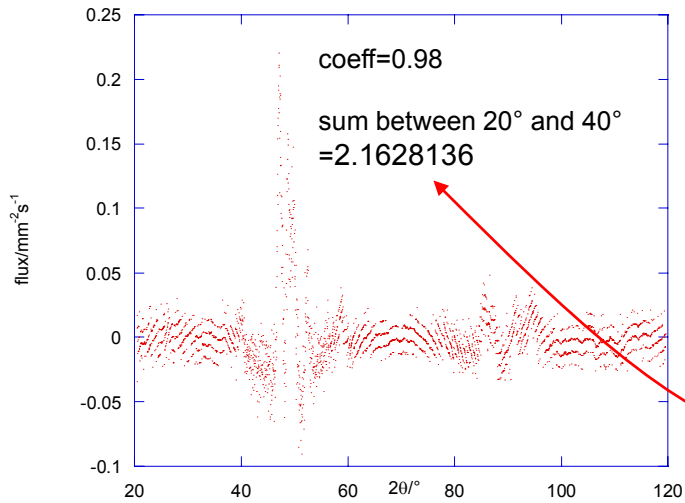
- Plot of the data from region 3, with 100% of the smoothed data subtracted:
- However, we know there are no crystalline peaks between  $20^\circ$  and  $40^\circ$



⇒ the sum of data values in this region...

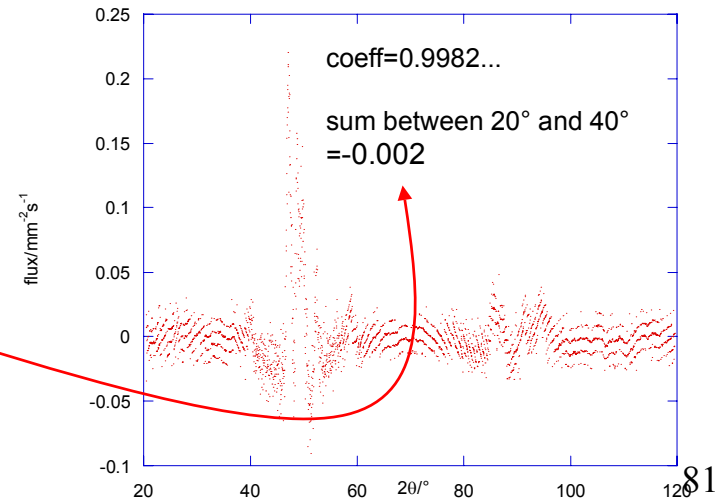
⇒ ...should be zero, if the correct scaling coefficient is used





**A coefficient of 0.98 underestimates the volume percentage of crystallite material...**

**...whereas one of 0.992 is getting closer to making the sum zero, as required**



- From this we conclude that the sample contains a volume fraction around  $(1 - 0.9982) \approx 1\%$  of material that isn't glassy  
     $\Rightarrow$  region 3 contains about 1% crystals by volume
- BUT is this accurate?

# LIGO Physical Property Measurements of Glassy Metals

Michael Hall

Six Month Intern  
Drexel University



Measuring Thermal  
Properties

Valerie Cervantes

High School Student  
Mayfield Senior School



Data Processing

# The Quantum Design Cryostat

- Temperature Range: 1.9K – 400K
- Pressure:  $8.9 \times 10^{-5}$  Torr
- Samples: Metglas, MoRuB



# Thermal Transport - Theory

1. Generate Heat Pulse
2. Measure  $\Delta T = T_{\text{hot}} - T_{\text{cold}}$
3. Estimate Corrections
4. Calculate Thermal Conductivity!

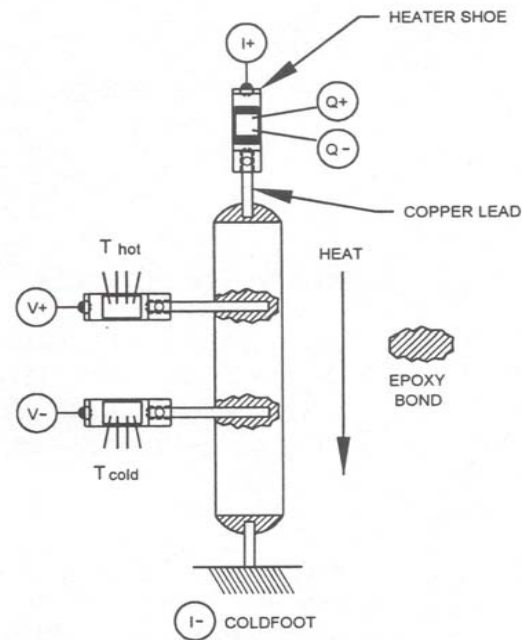
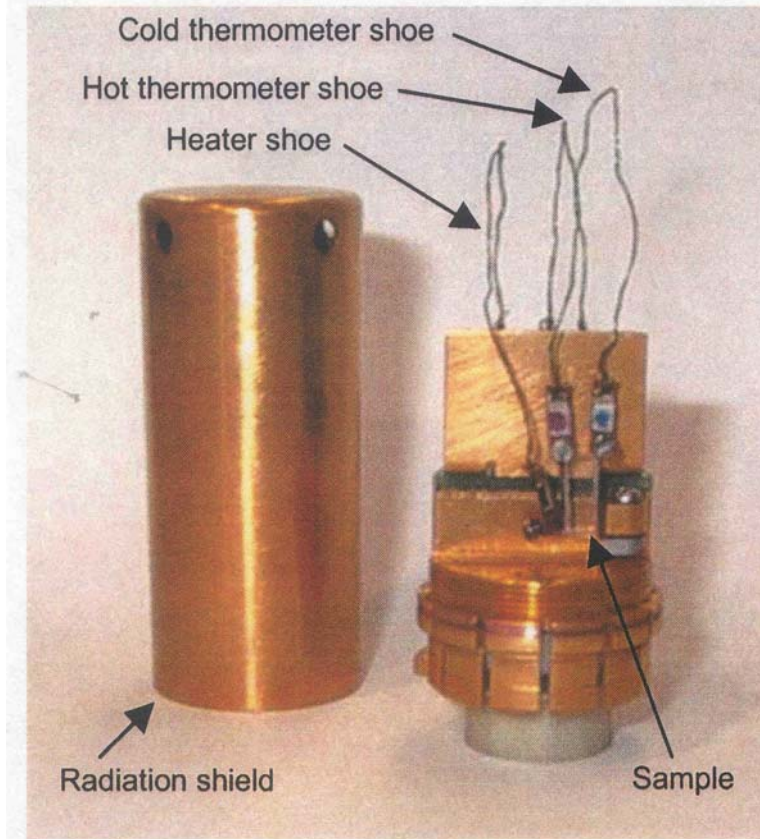


Figure 1-1. Thermal and Electrical Connections for an Idealized Sample

# Thermal Transport - Hardware



Thermal Transport sample puck with radiation shield

# The Cryostat – HiVac System

- HiVac Pressure:  $9 \times 10^{-5}$  Torr
- Many Problems!!
- Result: Quantum Design is Redesigning Cryopump

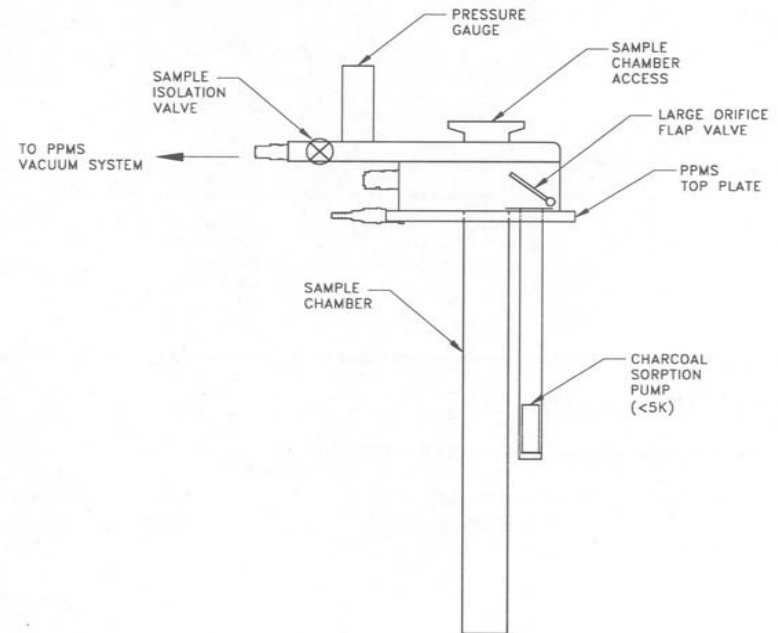
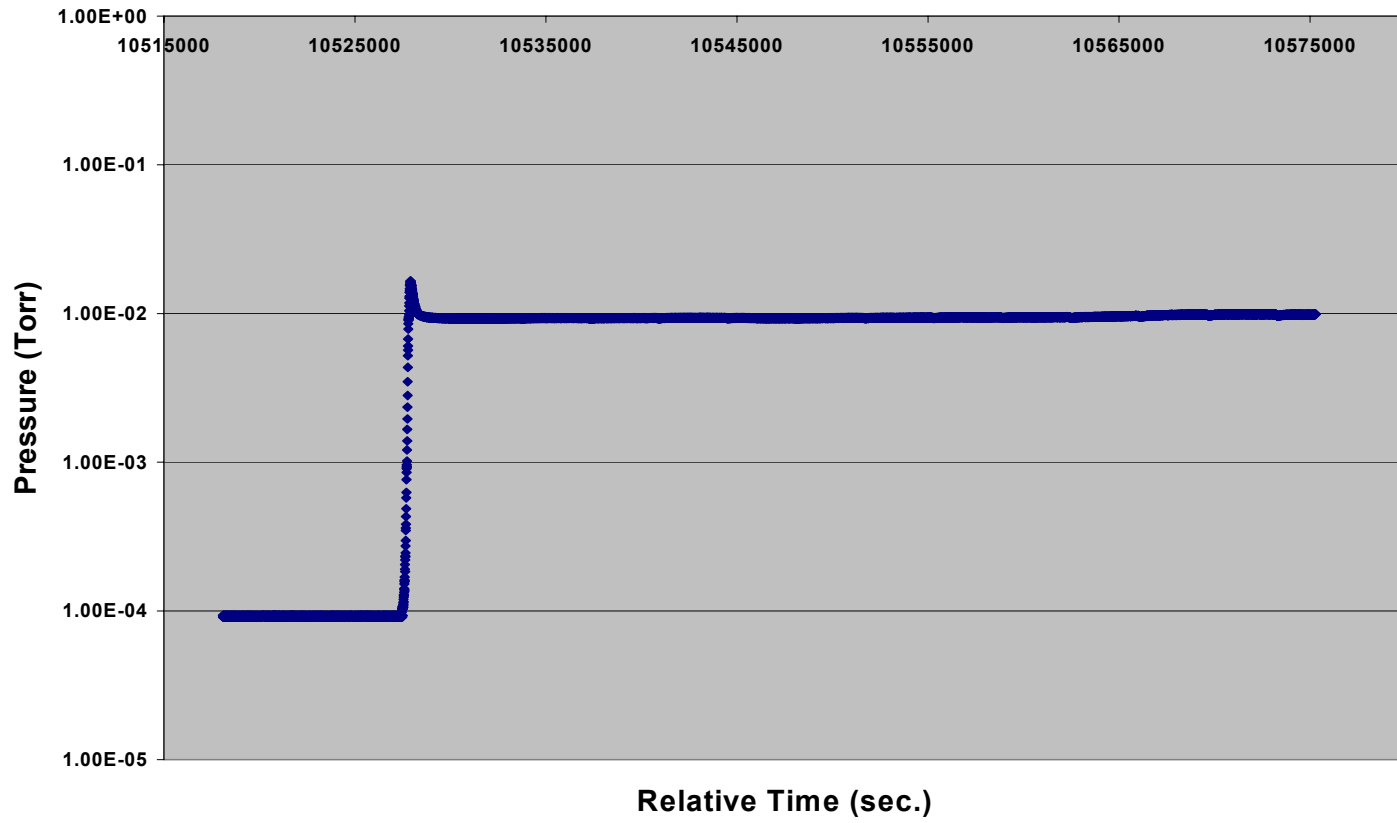


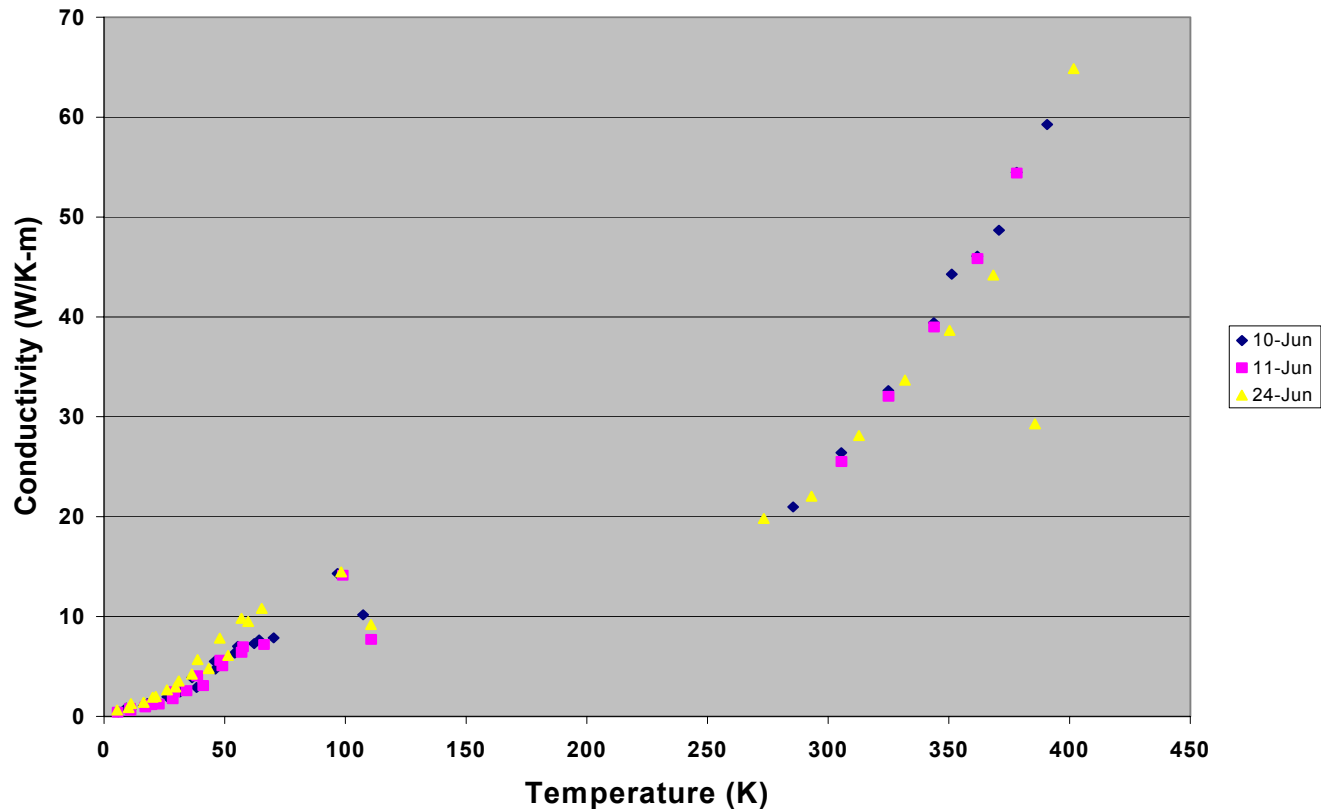
Figure 1-1. Cryopump Assembly

# Cryopump Failure

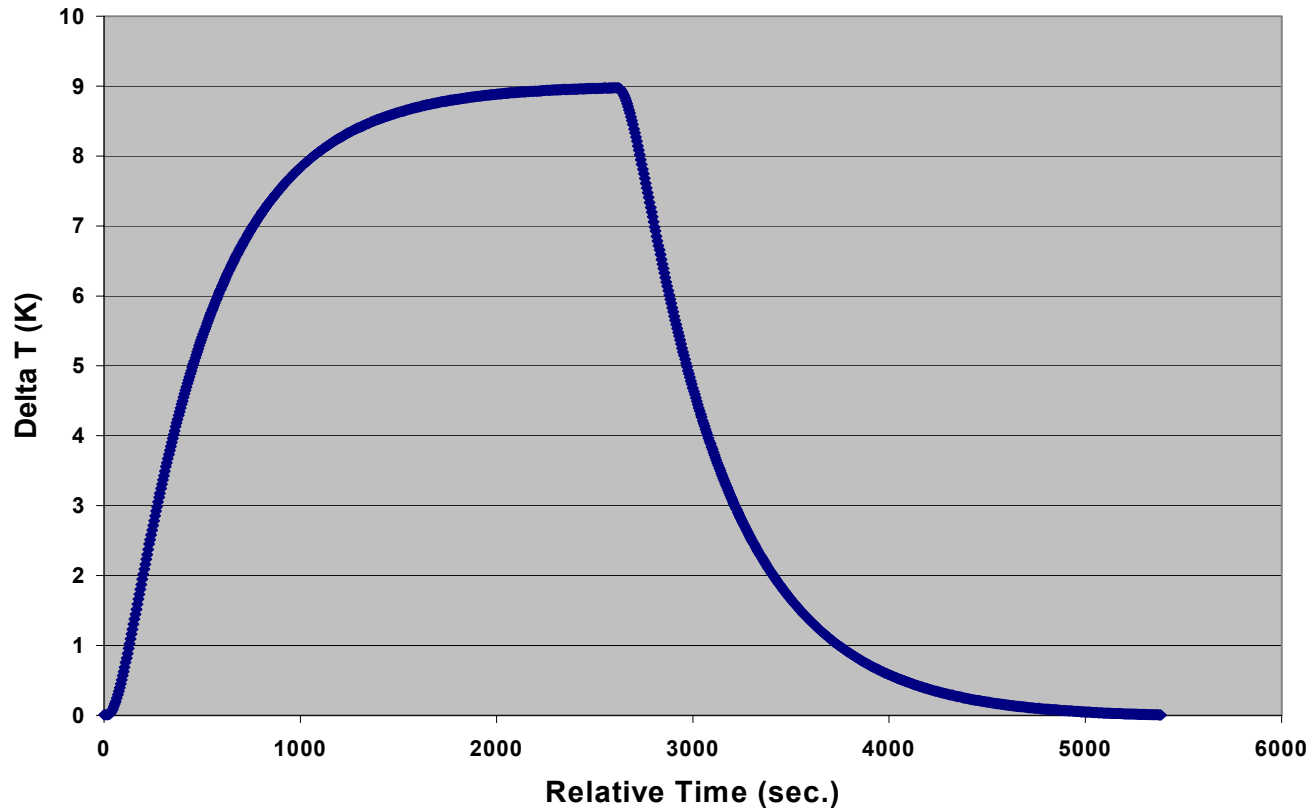




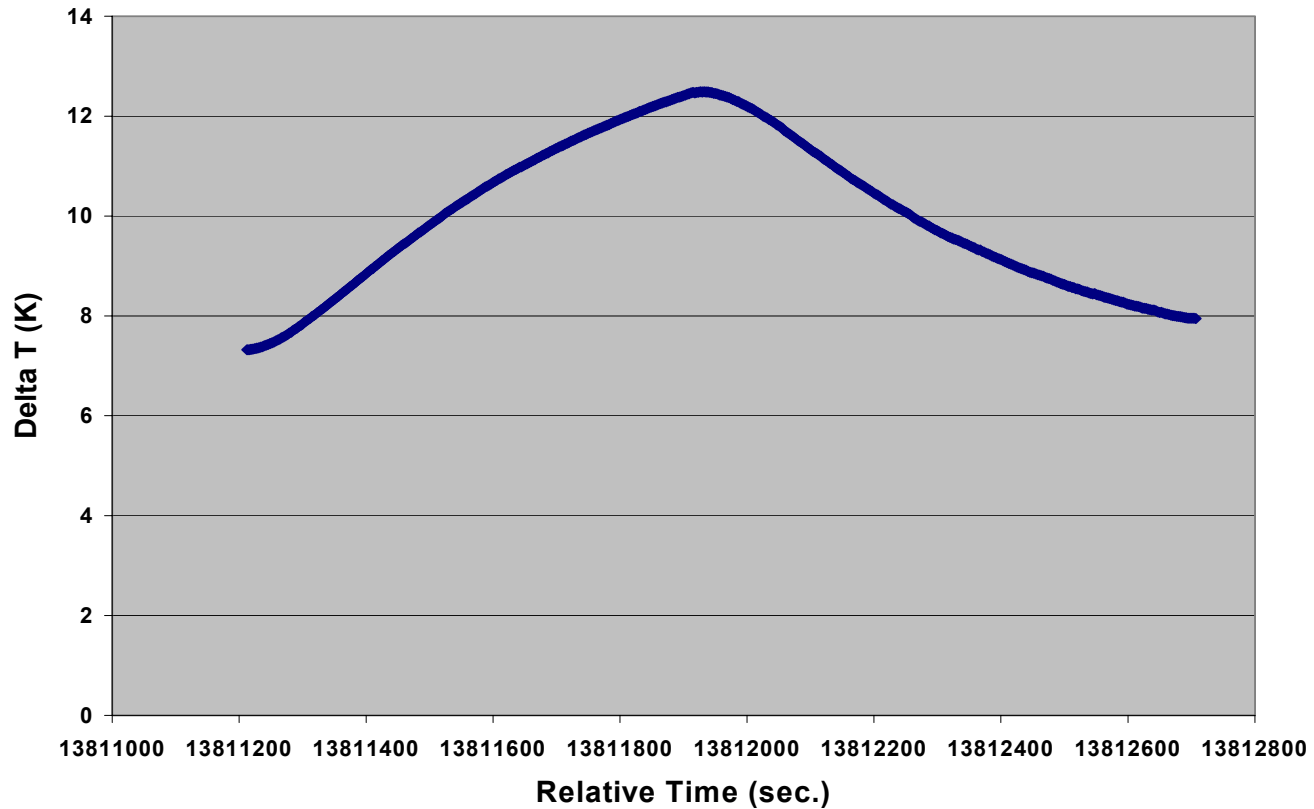
# Continuous Measurement



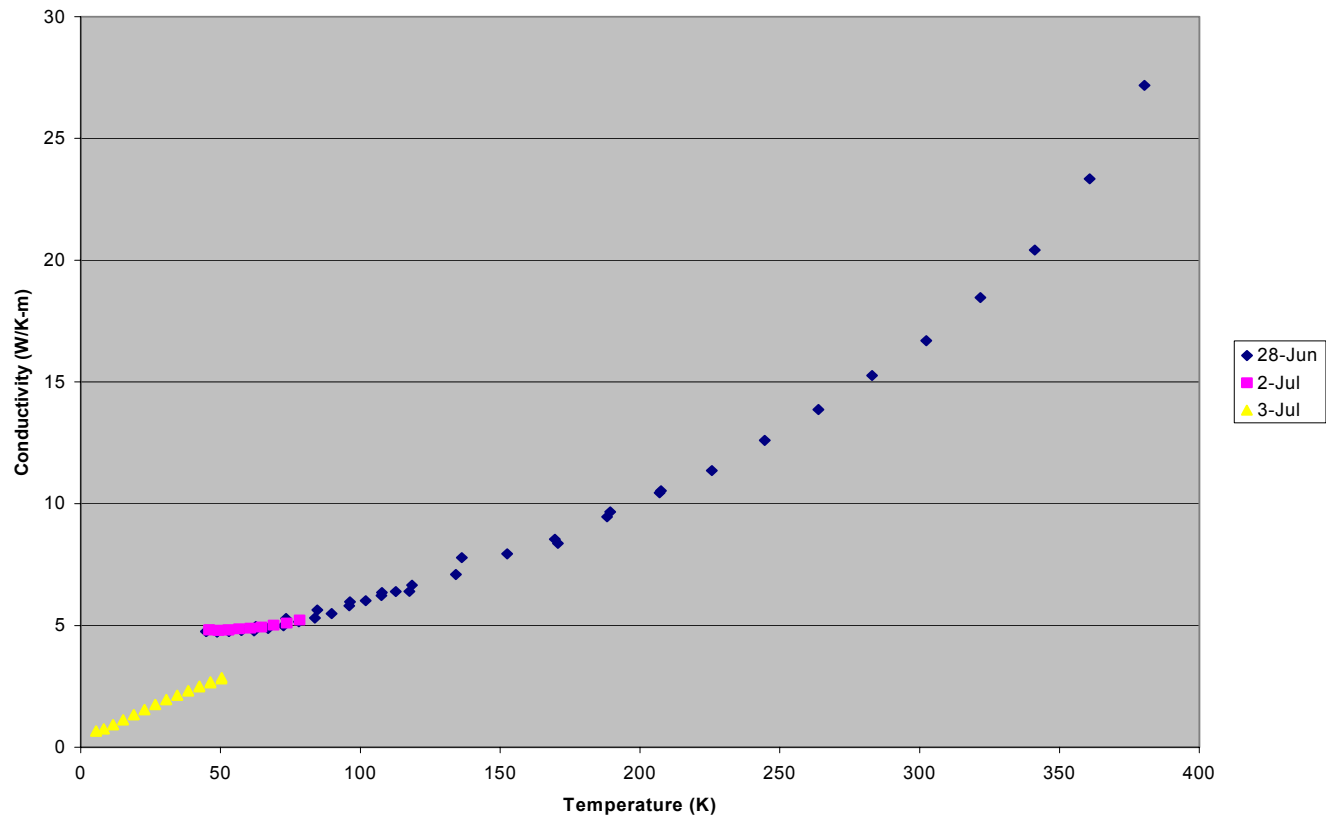
# Ideal Data Acquisition



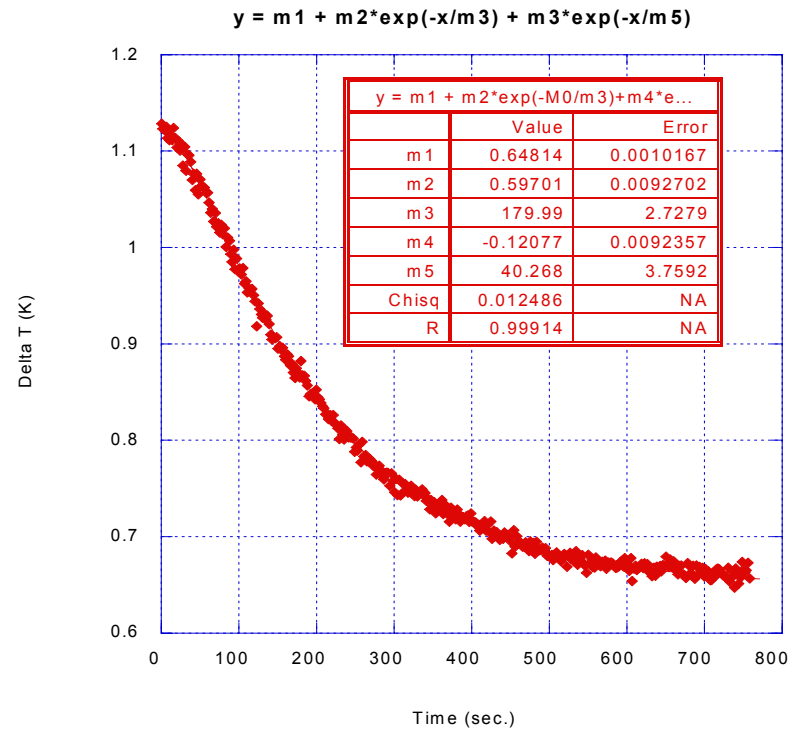
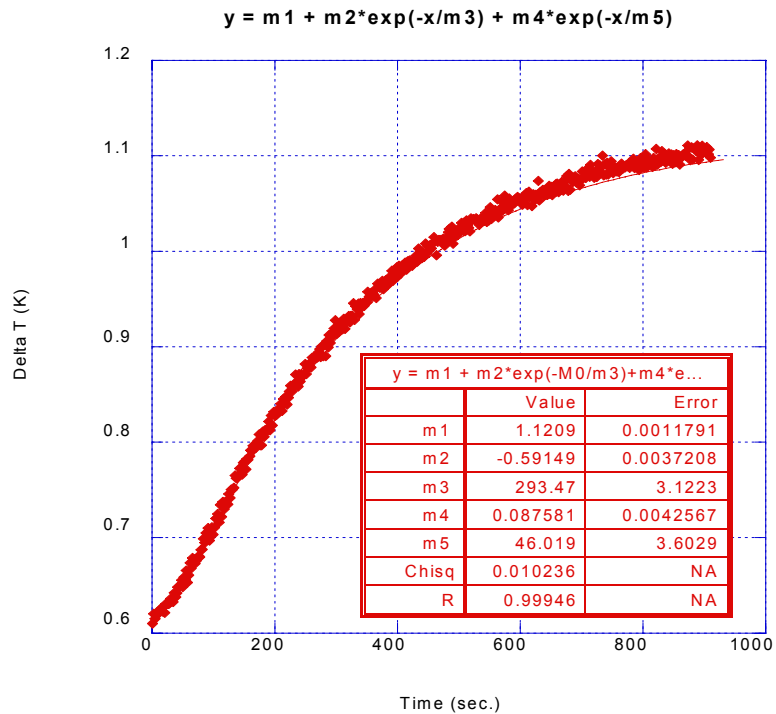
# Poor Data Acquisition



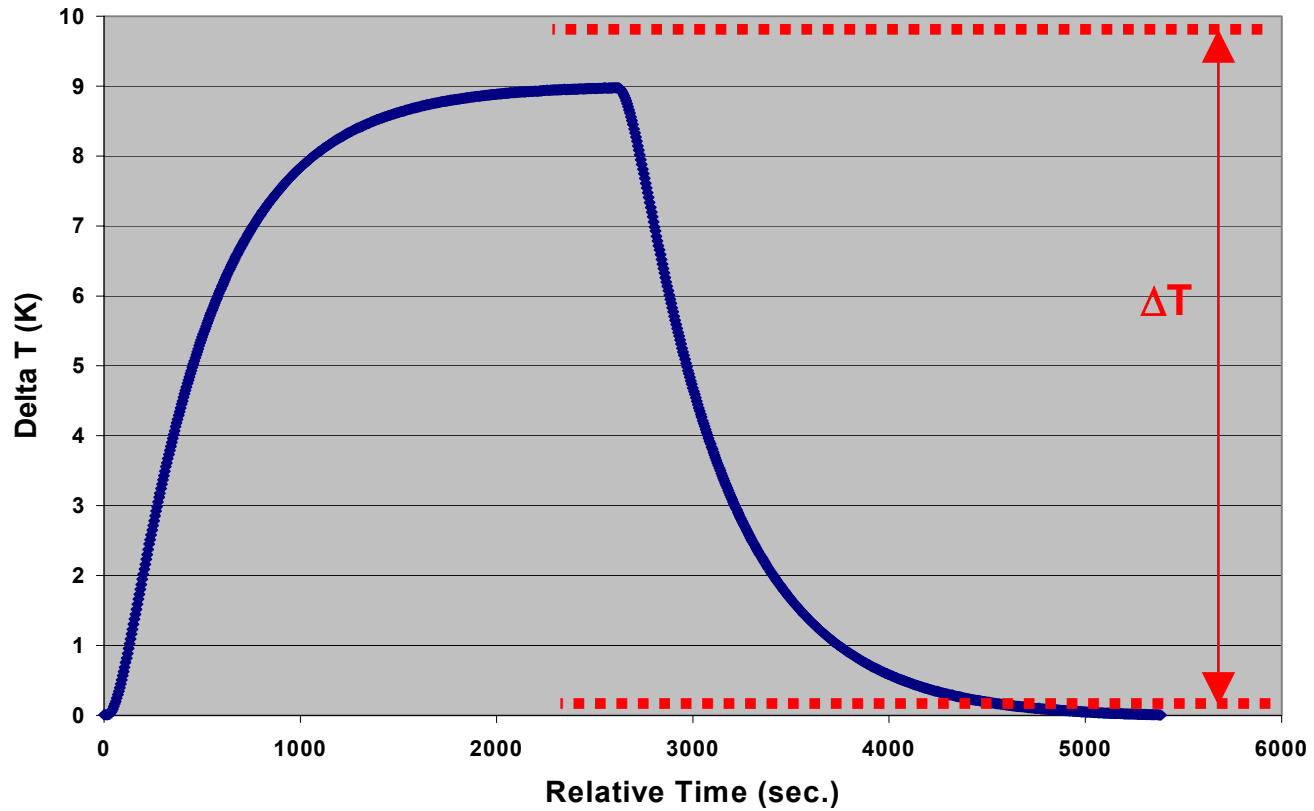
# TTO – Stable Measurement



# Data Processing



# Data Processing



# Thermal Transport Formulas

---

- Conductance
- $K = P/\Delta T$
  
- Conductivity
- $\kappa = [(P_{\text{IN}} - P_{\text{RAD}})/\Delta T - K_{\text{WIRES}}](L/A)$
  
- $P_{\text{RAD}} = \sigma\varepsilon(S/2)(T_{\text{HOT}}^4 - T_{\text{COLD}}^4)$
  
- $K_{\text{WIRES}}$  is Measured Directly!

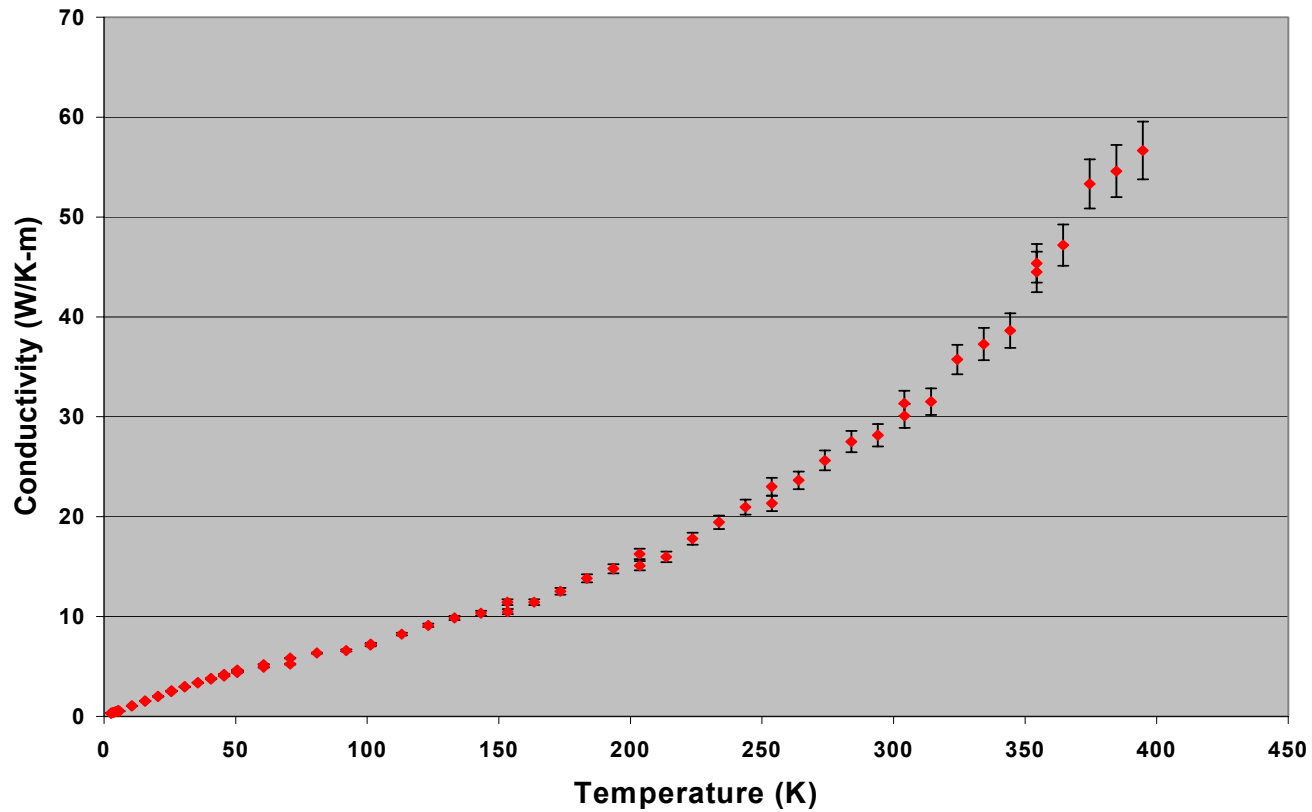
# Measurement Adjustments

---

1. Stable Measurements
2.  $\Delta T \sim 0.5K - 1.5K$
3. Measure Parasitic Conductance
4. Manual Data Processing



# Thermal Conductivity



# Heat Capacity - Theory

1. Generate Heat Pulse
2. Measure RC
3. Estimate Corrections
4. Calculate Heat Capacity!

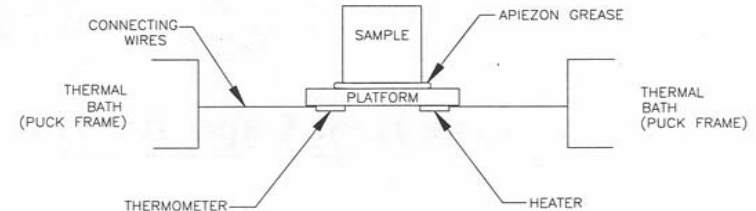


Figure 1-1. Thermal Connections to Sample and Sample Platform in PPMS Heat Capacity Option

- $\Delta T = T_{\infty} \exp[-t/R_8(C_P + C_G + C_S)]$

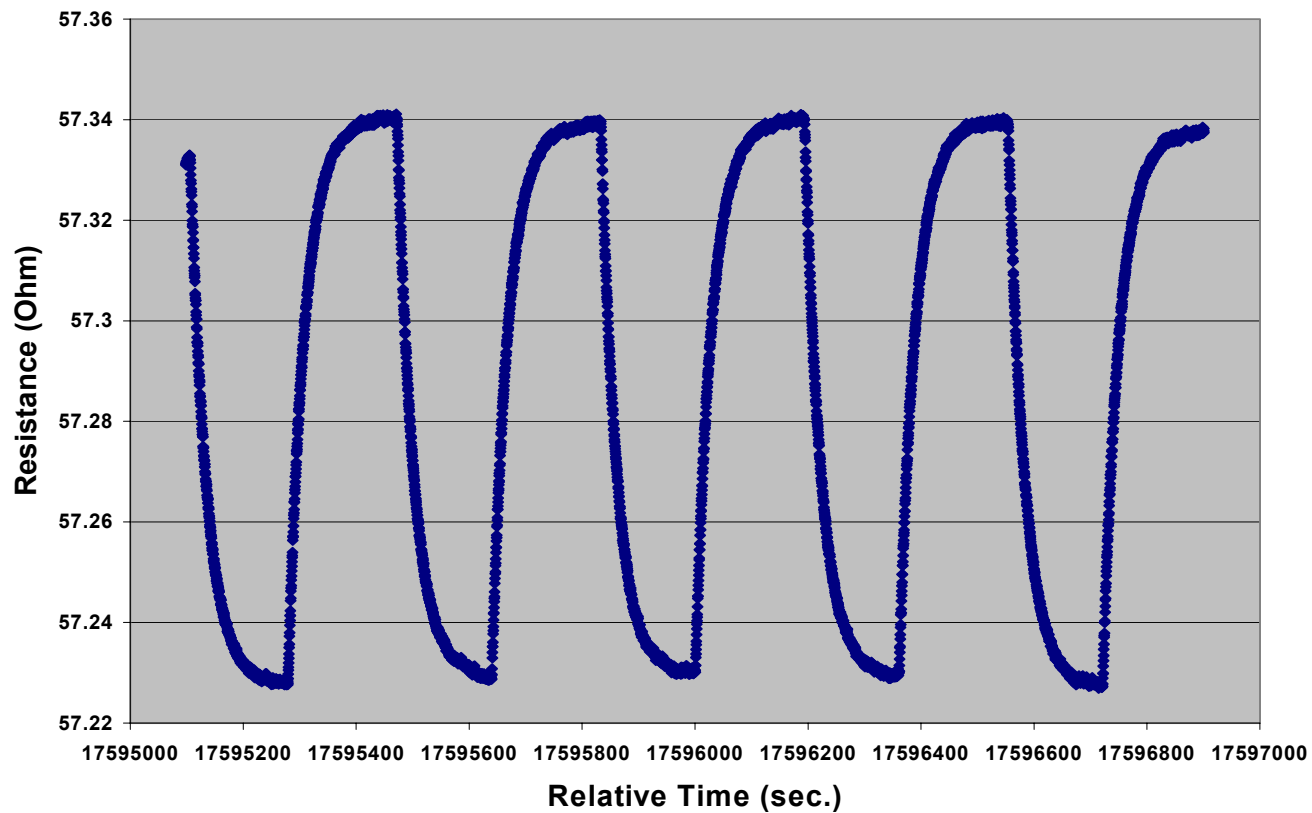
# Thermal capacity sample measurements

- Calorimeter puck





# Heat Capacity – Sample Data



## What is being done?

---

- Measure physical properties

- Thermal expansion coefficient

Continuously between 300 and 500 Kelvin (if necessary  
between 70 and 500 Kelvin)

- At Caltech's metallurgy extensometer

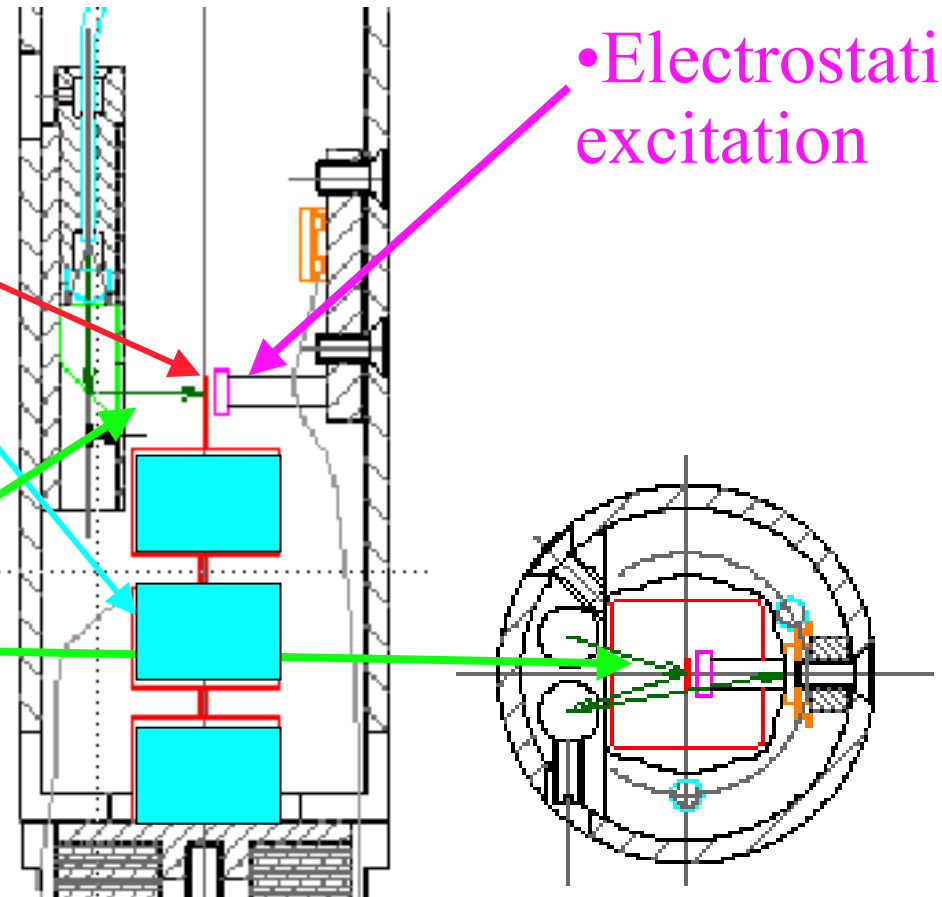
## What is being done?

---

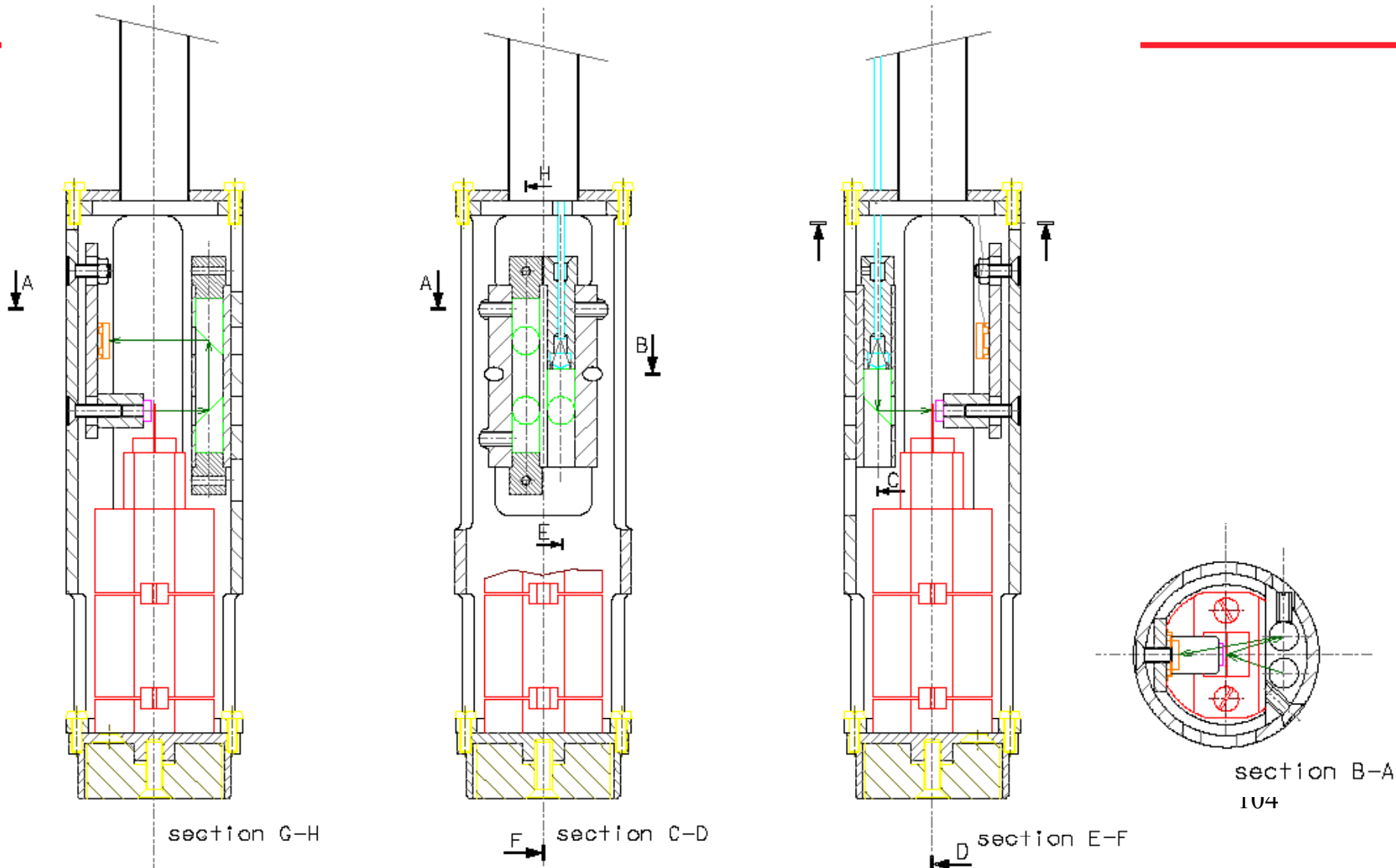
- Measure diving board Q-factors of samples
- At all temperatures between 2 and 400 Kelvins
- In Caltech's programmable cryostat
- Temperature in wide thermal range will allow identification and location of possible Debye peaks and estimate impurity ill effects

# Measure reed Q-factors

- **Reed** mounted on an **isolation stack** to isolate it from cryostat dissipation.
- **Optical lever** readout of ringdown



# Measure reed Q-factors

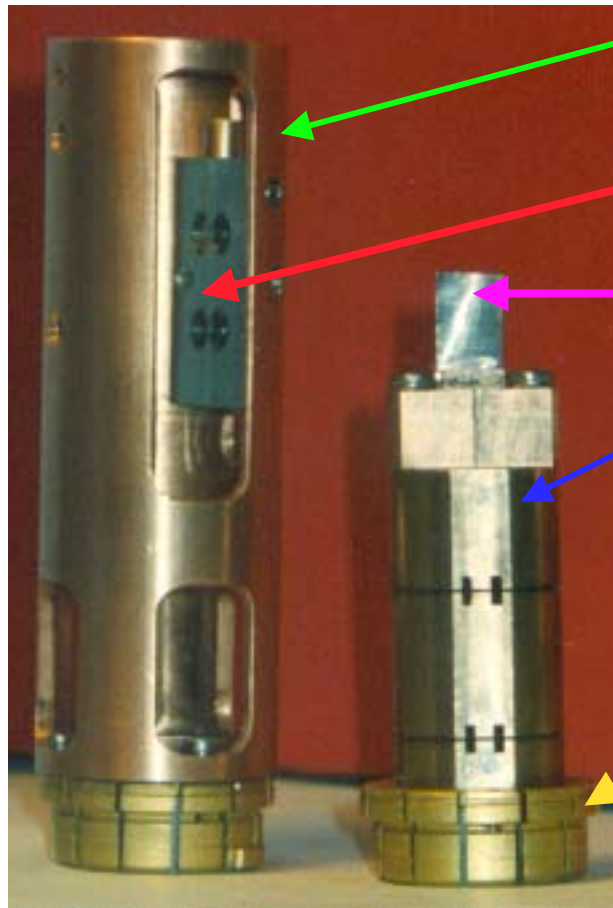




# Q Factor Measurement



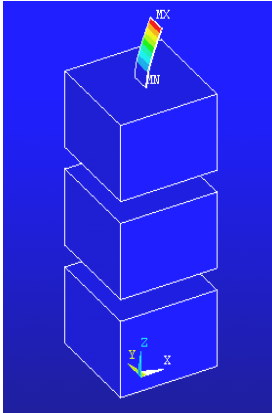
# Measure reed Q-factors



Cryo puck case,  
without Q-factor probe  
with periscope housing

Test reed on  
Q-factor probe on puck

# Measure reed Q-factors



	Frequency (Hz)	Mode Shape
<b>F<sub>1</sub></b>	<b>66</b>	Rotation around <b>y</b> , at lowest blade
F <sub>2</sub>	129	Rotation around <b>x</b> , at lowest blade
F <sub>3</sub>	250	Twist around <b>z</b> , at lowest blade
<b>F<sub>4</sub></b>	<b>388</b>	2 nodes oscillation around <b>y</b>
F <sub>5</sub>	694	2 nodes twist around <b>z</b>
F <sub>6</sub>	751	2 nodes oscillation around <b>x</b>
<b>F<sub>7</sub></b>	<b>896</b>	3 nodes oscillation round <b>y</b>
F <sub>8</sub>	1001	3 nodes twist around <b>z</b>
F <sub>9</sub>	1730	3 nodes oscillation around <b>x</b>
<b>F<sub>10</sub></b>	<b>3222</b>	Rotation around <b>y</b> of the reed

(10 micron thick reed)

## What to be done next?

- **Demonstrate feasibility** of employing Glassy Metals to fabricate mirror suspensions with record Q-factor
- **Ingredients**
  - Suspension rigid structure carved by EDM
  - Glassy metal Flex joints brazed to the rigid structure
  - Flex joint structure brazed to a wire
  - Hook bonded to a ledge in the mirror

---

QuickTime™ and a  
Photo - JPEG decompressor  
are needed to see this picture.

## What is being done?

---

- Can use glassy metal **Wire with “low” Q** to eliminate violin resonance problems

---

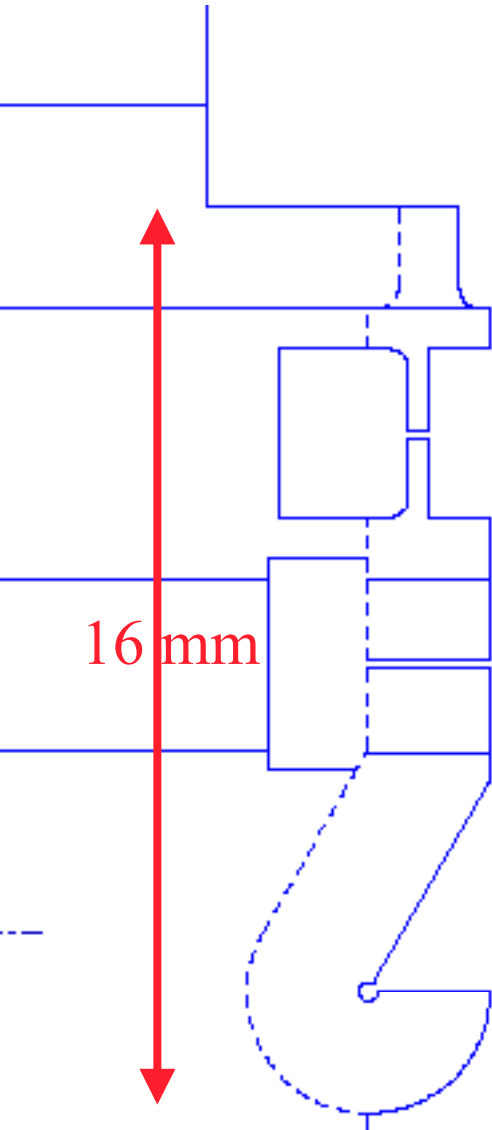
QuickTime™ and a  
Photo - JPEG decompressor  
are needed to see this picture.

## Fabricate the Flex Joint

EDM carve **half of the Flex Joint** structure out of a **single piece of material**

The Flex Joint structure will be **finished at the very end of the process** by **cutting the dashed lines**

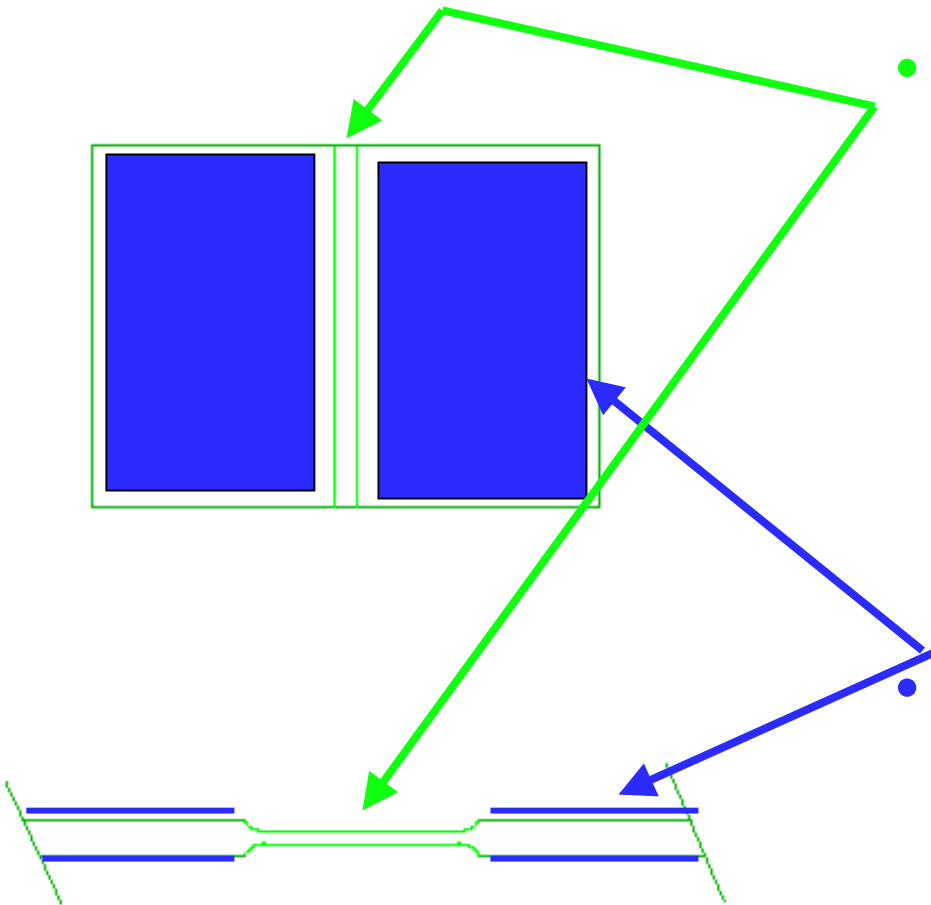
**All the surfaces on which to braze the flex joint are aligned by birth!**





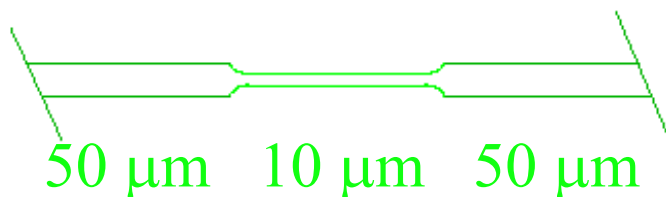
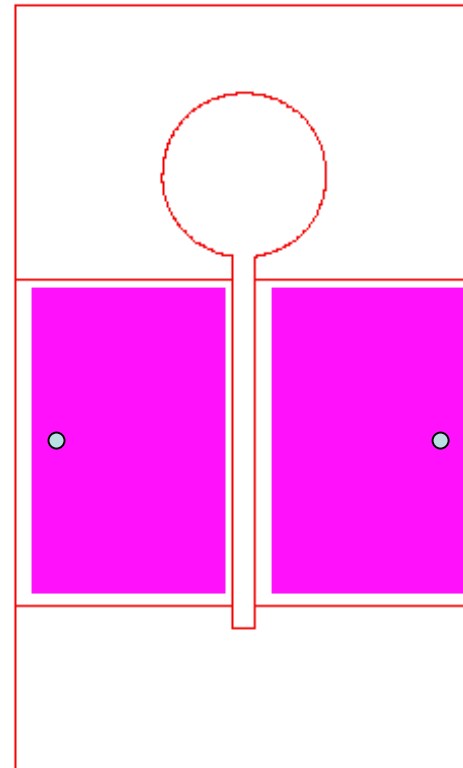
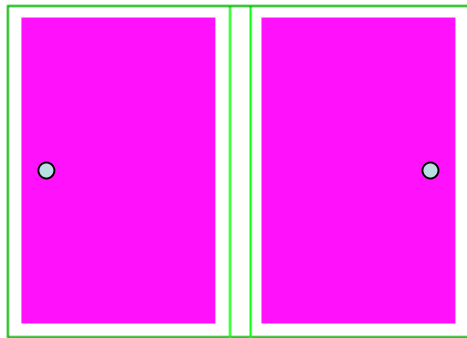
## Fabricate the Flex Joint

- The Flex Joint is obtained from a ribbon of glassy metals by thinning it from 50 to 10  $\mu\text{m}$  by through-mask electrochemical micromachining (IBM patent)
- Braze is deposited here

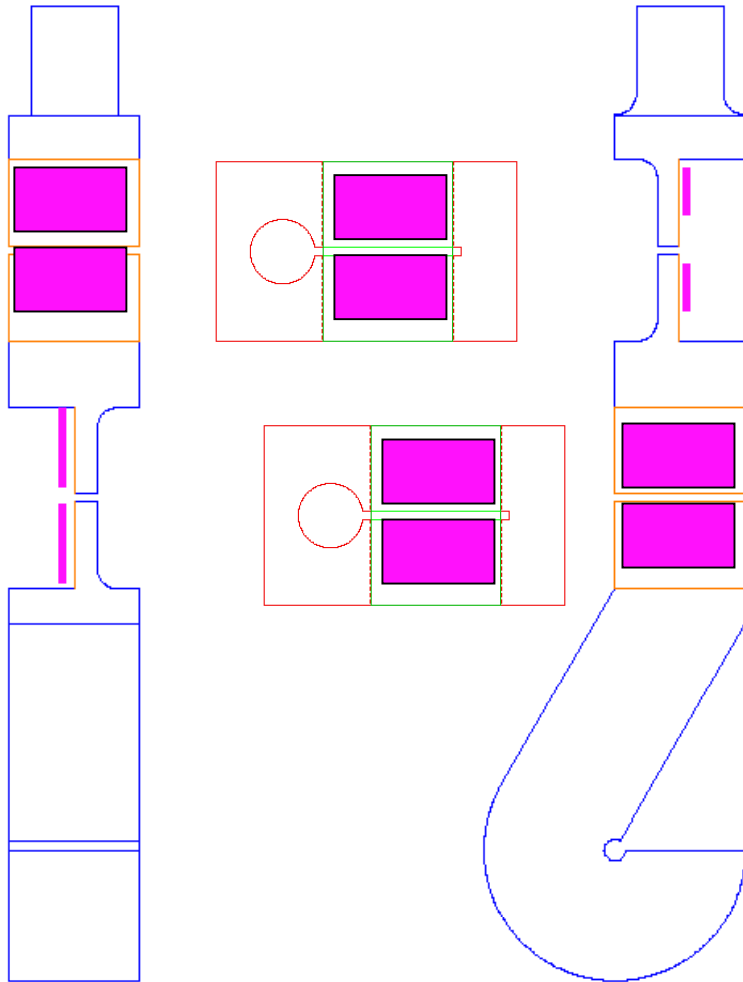


# LIGO Fabricating the Flex Joint

- The two Flex Joint covers are pre machined in the form of a "Cavalier", which houses and position the flex joint. The cavalier will be separated in the two covers in the final steps

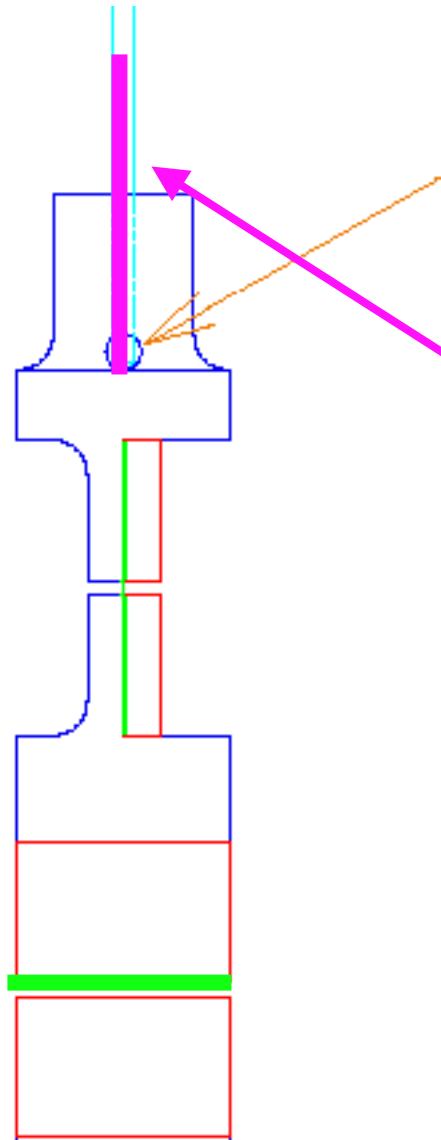


## Fabricate the Flex Joint



- For positioning the flexure two **cavaliers** encasing a flex joint are mounted **straddling the main structure**
- (Note: The flex joint structure is still monolithically attached to the mother structure)

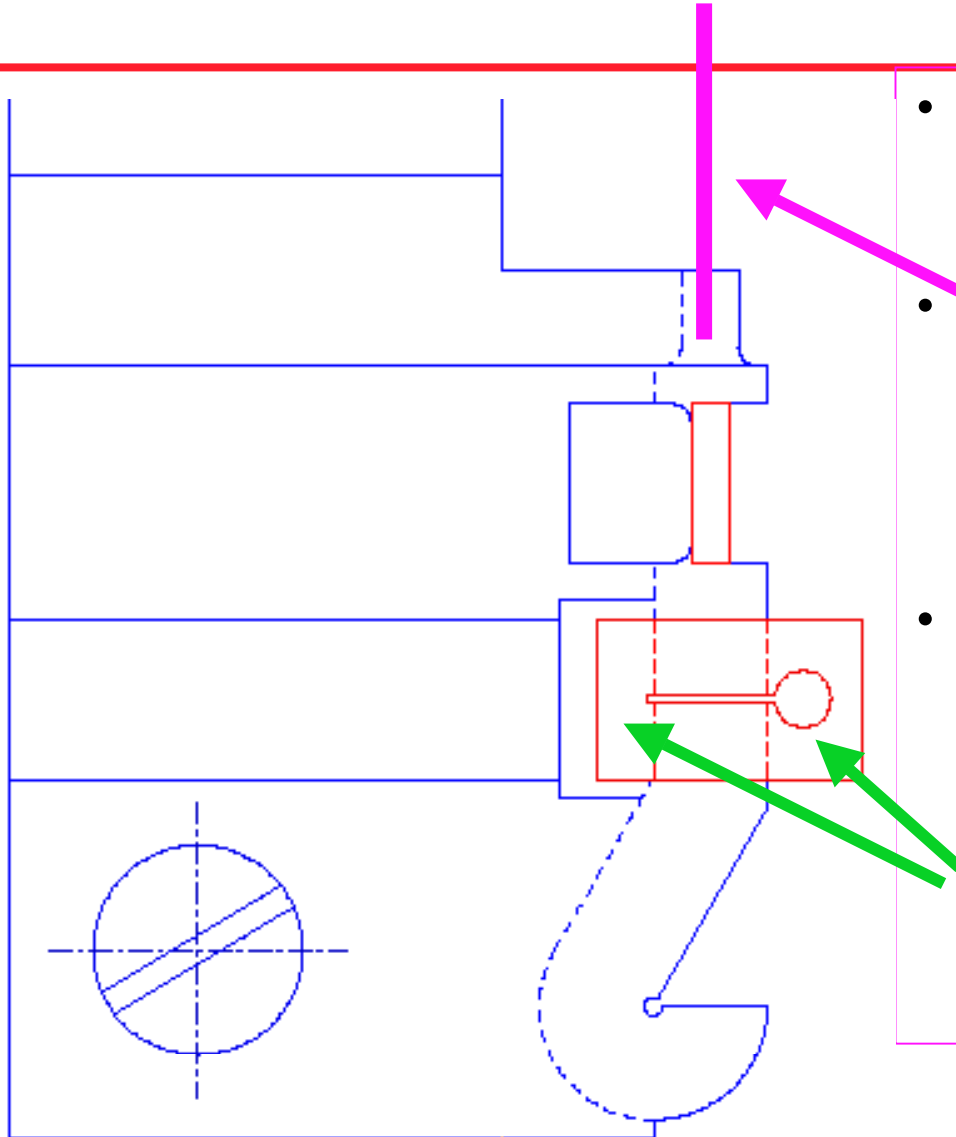
## Fabricate the Flex Joint



Braze material

- The suspension wire is positioned to be brazed as well

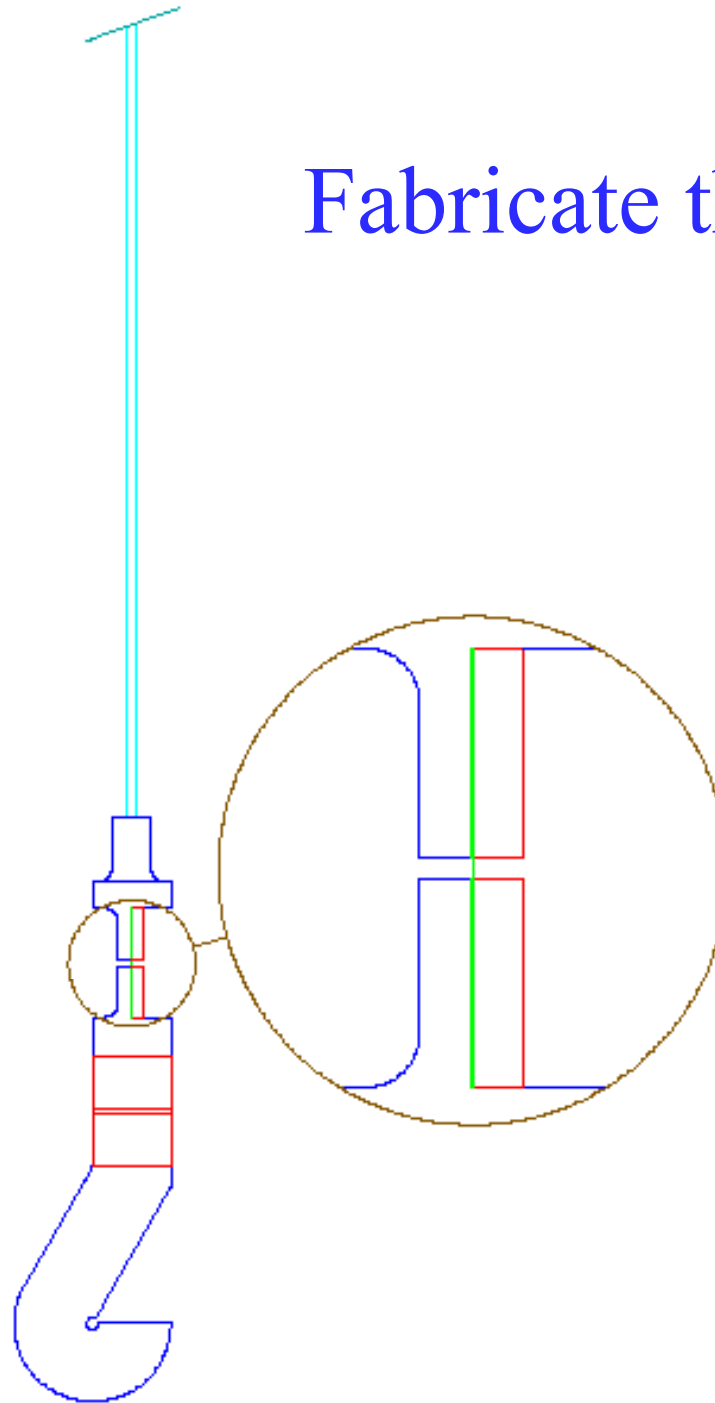
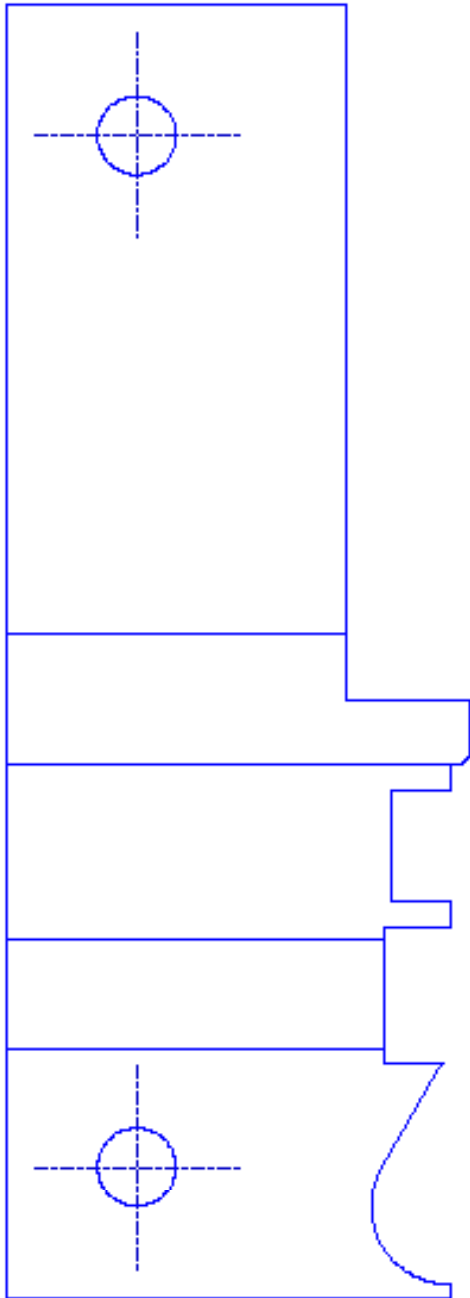
## Fabricate the Flex Joint



- The flex joint structure, is now provided with the glassy metal suspension wire
- The thin flex joints, are still imprisoned by the cavaliers both are brazed together by the baking process
- After brazing the ears of the cavaliers are EDM chopped off before separating the structure from its mother plate

# Fabricate the Flex Joint

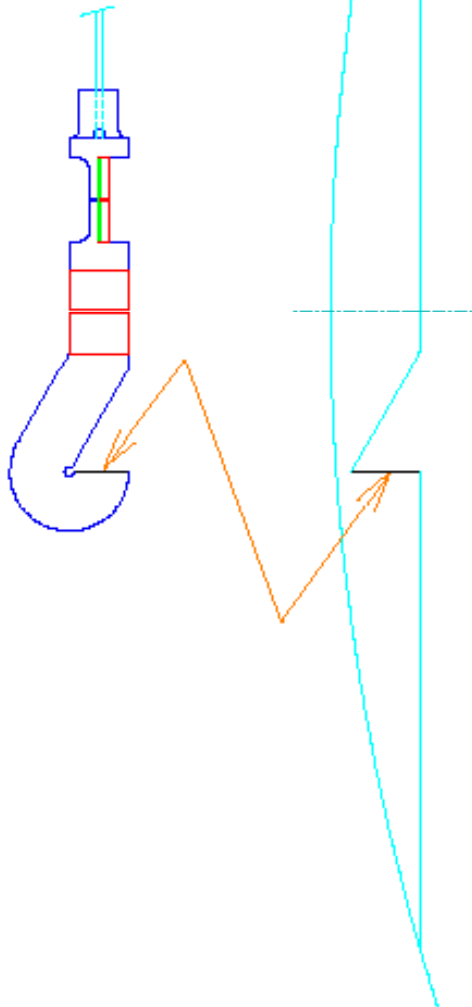
---



The finished flex joint is cut away and ready for attachment to the mirror

## Fabricate the Flex Joint

---



- The mating surfaces of the flex joint and of the mirror's ledge are indium coated to provide an excess-noise-free connection

## Why using ledges

---

- The use of **ledges** and low temperature brazing **eliminated all shear efforts**
- Can be **assembled and disassembled** by simply warming up the indium



## What else needs to be done?

---

To obtain ledge attachments on the mirrors

- Silicate bond ledges to mirror sides

or

- Ultra Sound machine (new advances)
  - Machine ledge-in-groove in sides of mirrors

## What more to be done?

---

- Demonstrate feasibility of attachments to mirrors without significant loss of mirror Q-factor
- Will use assistance from Kenji Numata's setup to measure the localized loss of Q-factor and from Kazuhiro Yamamoto to evaluate its effect on mirrors' performance

To physically measure suspension  
Q-factors ( $>10^8$ ) with macroscopic mirrors

---

Propose to use Opposite philosophy of all existing facilities

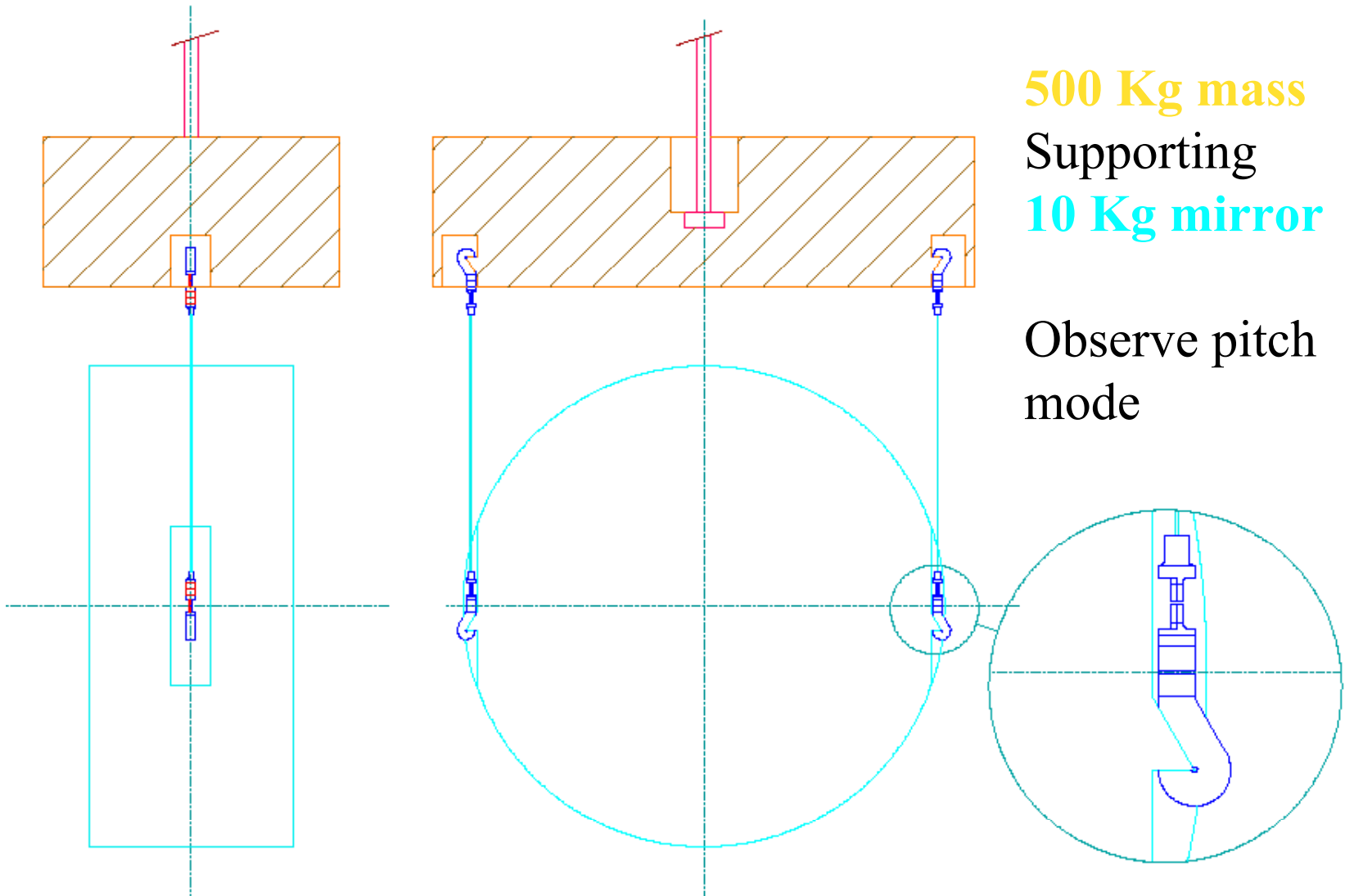
- Do not look for biggest available rock !
- Build softest available support to decouple ground losses from mirror suspensions under test
- Build mirror suspension test facility taking advantage of existing Ad-LIGO test tower
- Measure pendulum or, better, pitch mode

**LIGO**

# What is being done?

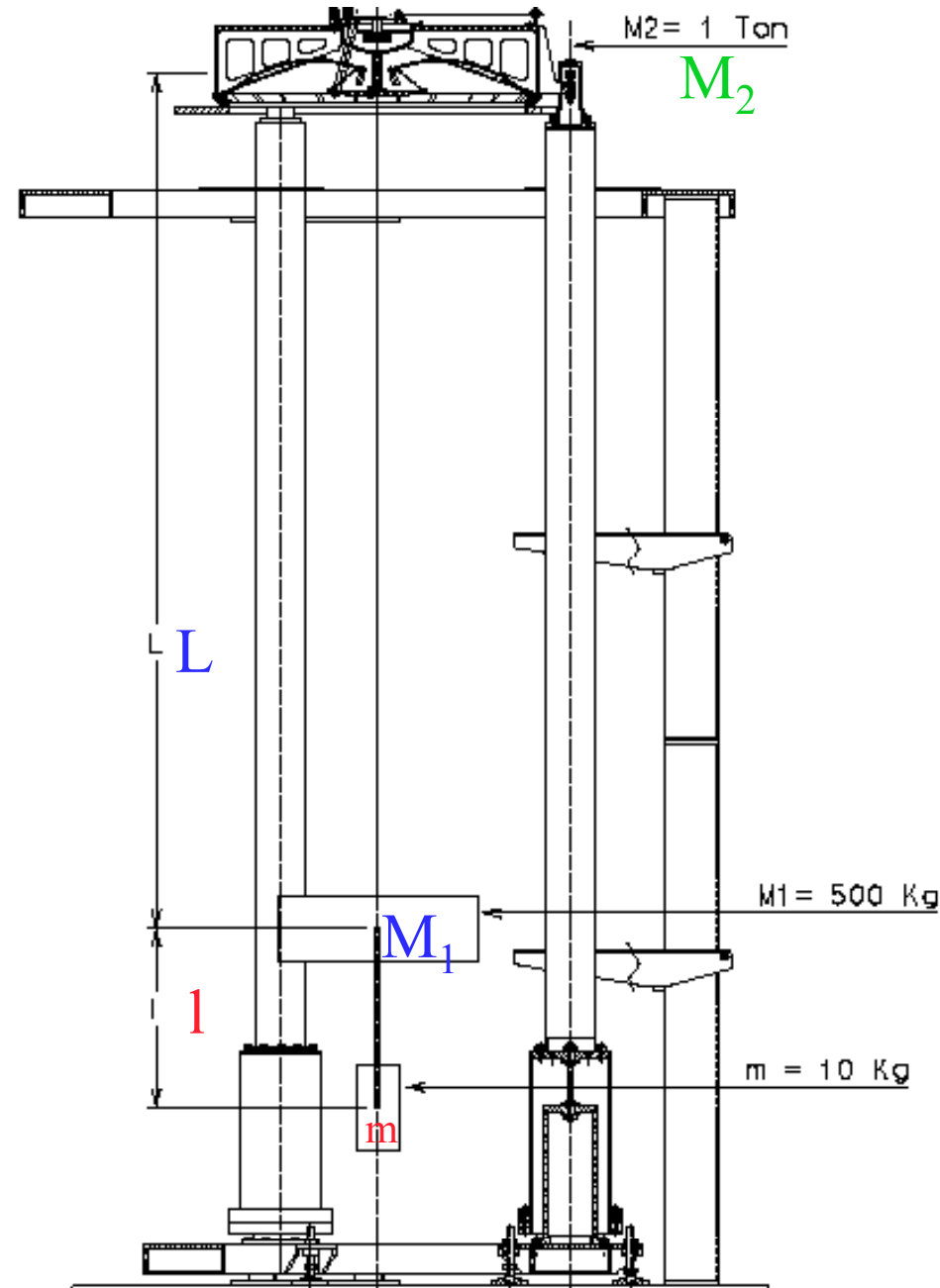
**500 Kg mass**  
Supporting  
**10 Kg mirror**

Observe pitch  
mode



## What is being done?

- Losses to ground decoupling factors:
- Measure pitch decay
- $\Delta\theta/x$  (depending on balance)  $M_2$  50?
- $\Delta m/M_1$  10/500  $M_2$  50
- $\Delta m/M_2$  10/1000 100
- $\Delta l/L$  20/400 20
- IP isolation factor ?
- Total  $> 5 \cdot 10^6$
- Facility should be able to measure oscillation  $Q \gg 10^8$



What is being done?

